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## Ferroelectrics

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# Successive phase transitions in the MeLiBO<sub>4</sub> type crystals

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SUCCESSIVE PHASE TRANSITIONS IN THE MeLiBO<sub>4</sub> TYPE CRYSTALS

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<u>Abstract</u> The properties of pseudo-proper ferroelectrics of the CsLiMoO<sub>4</sub> group are reviewed. The structures of distorted phases and calorimetry of the crystals are discussed. The model of the order-disorder type transitions is proposed and solved in a cluster approximation. Comparison with experimental data is made.

#### 1. INTRODUCTION

Ferroelectric properties of the double alkaline molybdates and tungstates:  $CsLiMoO_4$  (CLM),  $CsLiWO_4$  (CLW) and RbLiMoO\_4 (RLM) were found and studied recently in this laboratory <sup>1-3</sup>. CLM and CLW crystals at room temperature belong to the cubic acentric space group  $G_0 = T_d^2 = F\bar{4}3m$  and show a remarkable piezoelectric effect and second optic harmonic generation. Their structure is based on a frame of corner-linked  $MoO_4$  and  $LiO_4$ tetrahedra, and topologically related to the well-known structure of the high cristobalite form of  $SiO_2$ . Large alkaline cations occupy one half of the 12-coordinated Laves polyhedra of the frame. In the course of the  $G_0$ -structure determination <sup>4</sup> it has been noticed that Cs-O distances in this polyhedron for CLM substantially exceed the sum of their ionic radii and the thermal parameters of oxygens are abnormally large if these atoms are located on the 3-fold axes of the structure (16-fold positions). These thermal parameters and R-factors were remarkably lowered when the statistical disordering of oxygens on six general positions around 3-fold axis was included in the model of structure (96-fold positions,  $\Delta r_0$  0,26 Å). The analogous conclusion was independently made in the course of the G<sub>0</sub>-structure study for CLW <sup>5</sup>.

The existence of statistical disorder in  $G_0$ -phases initiates both our study of the probable structural phase transitions (PT's) in CLM and CLW below room temperature and efforts to synthesize the Rb- and K-isomorphs of the crystals. Cristobalite-like phase was found 7 in RLM above 400 K. Potassium-lithium molybdate and tungstate transform to the same structure above 630 K but the cubic phase is unstable and the crystals have the tridimite-like structure at room temperature  $\frac{6}{6}$ .

In this paper the previous results on the study of PT's, ferroelectric and elastic properties of the CLW, CLM and RLM crystals are shortly reviewed and then the new data on structures of the low temperature phases and the adiabatic calorimetry of the crystals are discussed in the connection with the statistical theory of PT's which was shortly presented in <sup>7</sup>. The results of the theory in the cluster approximation may be now compared with the experimental data.

#### 2. PHASE TRANSITIONS

Single crystals of CLM, CLW and RLM were grown by Bridgman method in the sealed platinum ampoules from the melt of corresponding stoichiometric substances which were prepared beforehand by means of the solid-phase synthesis from the anhydrous salts, e.g.  $Cs_2MoO_4$  and  $Li_2MoO_4$  for CLM<sup>4</sup>. Experimental samples were cut out from annealed boules (diameter ~10 mm, ~30 mm in lenght) which were optically clean and lightly coloured. X-ray method was used for the orientation of the samples. The dielectric, ferroelectric, elastic properties and the optical twinning below PT's were studied earlier <sup>1-3</sup>, Raman-spectrum and NMR data were presented in <sup>9</sup>, some calorimetric data in <sup>7,10</sup>.

The properties of the CLM and CLW crystals are closely related. Both crystals undergo the sequence of the two first order PT's  $G_0 G_1 G_2$  at temperatures  $T_{c1}=220$  K,  $T_{c2}=185$  K for CLW and 209 K and 171 K for CLM. These PT's proceed without multiplication of the formula unit number in the (primitive) unit cells and the following symmetry changes at PT's were determined 1, 3, 7:

$$\mathbf{F}_{d}^{2}(\mathbf{Z}=1) \rightarrow \mathbf{C}_{3\mathbf{v}}^{5}(\mathbf{Z}=1) \rightarrow \mathbf{C}_{s}^{1}(\mathbf{Z}=1)$$
(1)

Unit cell parameters are presented in Table 1, where F-centered unit cell (Z=4) is used for  $G_0$ -phases.

The deformed hysteresis loops were observed in  $G_1$ and  $G_2$  phases. The values of the spontaneous polarization (P<sub>s</sub>) were measured by the pyroelectric charge method <sup>1</sup> only for  $G_1$ -phases. Just above  $T_{g2}$  the values are equal ~ 0,4 and 0,07 (in 10<sup>-2</sup> C·m<sup>-2</sup>) for CLM and CLW respectively. In  $G_2$ -phases this method gives the lowered values of P<sub>s</sub> due to the complex twinning of the samples. Single domain samples were obtained only in the  $G_1$ -phases by applying dc electric field or mechanical stresses. The main peculiarities of the physical properties in both crystals are the absence of dielectric anomaly and the strong decrease of shear elastic stiffness  $c_{44}$  in  $G_0$  above  $T_{c1}$ . Due to their  $G_0$  structure both crystals in trigonal phases belong to the class of ferroelectrics where  $P_g \parallel [111]_c$  can not be reversed under biasing electric field but changes its direction to other 111 -type directions.

These data allow to conclude that both CLM and CLW are the pseudoproper ferroelectrics and ferroelastics. Thermodynamical theory of  $G_0 \rightarrow G_1 \rightarrow G_2$  transitions was published earlier<sup>1</sup>. The potential of the acentric  $G_0$ phase contains both even and odd order terms of threecomponent order parameter  $\gamma_i$  (i=1,2,3), which transforms as the irreducible representation  $F_2$ . The bilinear coupling of  $\gamma$  with elastic subsystem is much stronger than its interaction with the electric one. The properties of CLM were analysed in neighbourhood of  $G_0 \rightarrow G_1$  transition.

Rubidium lithium molybdate (RLM) undergoes three successive PT's at  $T_{c1}$ = 390 K,  $T_{c2}$ =380 K,  $T_{c3}$ =375 K. Above  $T_{c1}$  it is isomorphous with CLM and CLW. The twinning structure, dielectric properties and X-ray powder data show that the  $G_1$ ,  $G_2$  and  $G_3$ -phases belong to trigonal, orthorhombic and monoclinic systems respectively (see Table 1 and Section 3). The crystal is also ferroelectric below  $T_{c1}$ . The primitive unit cell of the room temperature  $G_3$ -phase contains two formula units (Z=2). The thermodynamical potential <sup>1</sup> can not be used for the description of PT with multiplication of the unit cell. Another order parameter should be included for RLM crystal. In Section 5 the statistical model is presented which can explain the main features of physical properties and the sequence of PT's in the whole group of crystals.

#### 3. STRUCTURES OF DISTORTED PHASES

In addition to X-ray powder analysis of distorted phases (Table 1) the atomic structures of the  $G_1$ -phase s CLW and  $G_3$ -phase RLM were determined. Automatic X-ray diffractometer Syntex Pī (MoK<sub>oC</sub> -line,  $2\Theta_g \Theta$  method with the variable scanning rate). Cylindrical sample of CLW in a capilliar was stressed mechanically and cooled by the nitrogen vapour stream below  $T_{c1}$ . Single domain state was optically controlled. 230 independent  $F_{hkl}$ at T=203 2 K were collected. Computing program Roentgen-80 was used for the solving of the structure.

It is worthwhile to compare the atomic coordinates of basic atoms for  $G_0^{5}$  and  $G_1$ -phases of CLW in the same presentation (R3m, hexagonal unit cell:  $a_h =$ 5,904 ,  $c_{h}=14,462 \text{ Å for } G_{0} \text{ and } a_{h}=5,836$  ,  $c_{h}=14,322$ for G1), Table 2. The comparison shows that all cations have nearly unchanged atomic coordinates in  $G_1$  in respect to G ones. The main changes take place in WO<sub>4</sub> tetrahedron. The W-O distance along 3-fold axis increases on 0,05 Å and three other bonds slightly decrease. It is also seen that both in  $G_0$  and in  $G_1$  (models I in Table 2) the isotropic thermal parameters of oxygens are abnormally large. Anisotropic thermal ellipsoids of O-atoms have a tablette-like form, their short axes are parallel to W-O-Li bonds. R-factor is 0,083 for the "ordered" model I of G1-phase. Model II for both phases corresponds to disordered oxygens, they were shifted from their mean positions of Model

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TABLE 1 The main crystallographic data

CeLiMoO <sub>4</sub> a <sub>0</sub> =8,318(4) Z=4	CaLiWO <sub>4</sub> T <sup>2</sup> , a <sub>0</sub> =8,250(4) Z=4	లి	RbLiMoO <sub>4</sub> T <sub>d</sub> (?), a <sub>0</sub> =8,115(6) Z=4
812(5) 98(8)	c <sup>5</sup> , α=b=5,836(3) c=14,322(8) Z=3	5	c <sup>5</sup> y, <b>e</b> =b=5,740(3) c=14,112(8) Z=3
		с ц	c <sub>2V</sub> , a=5,739(3) b=5,751(3) c=8,110(6) Z=2
	In	ີຍີ	I111/b
• •	<b>a =</b> 5,85(3) b = 5,86(2)		a = 10,018(2) b = 11,272(2)
) (8)	c = 8,34(5) = 90°48'(8)		c = 5,702(1) = 125,48'(2)
	2=2		2=4

I in m-planes. In this case each oxygen is statistically distributed on three positions. Disordered models II lead to the definite reduction of R-factors (R=0,062 for  $G_1$ ) and to the "normal" isotropic thermal parameters of atoms (Table 2).

So we can conclude that a disorder of oxygens taking place in  $G_0$  remains in  $G_1$ -phase and the trigonal symmetry of the structure is preserved due to dynamical disordering of the  $WO_4$ -tetrahedra around 3-fold axis. It can be supposed that the main increment to  $P_s$  in  $G_1$  is determined by the small dipole moments of slightly distorted  $WO_4$ -groups; according to NMR-data <sup>9</sup> the distortions of Li-environment in  $G_1$  are extremelly small.

The froasing of tetrahedra in the shifted positions "in phase" leads to the monoclinic structure of  $G_2$ -phase. The straight determination of the  $G_2$ -structure is a difficult problem due to superposition of several systems of twins in a sample. The efforts to detwin the  $G_2$ -phase were unsuccessful yet.

The structure of RLM ( $G_3$ -phase) was published earlier <sup>7,8</sup>. After three phase transformations the structure maintains the cristobalite-like tetrahedral frame. The displacements of Rb (0,13 Å) and Li (0,06 Å) in respect to their  $G_0$ -positions take place. Slightly distorted tetrahedral MoO<sub>4</sub>-groups as a whole turn on angles 20° around c and b axes of monoclinic unit cell (directions [011] and [017] in  $G_0$ ). The tilts of tetrahedra around c are "in phase". Their antiphase tilts around b lead to doubling of the unit cell. The thermal parameters of all atoms have the normal values and we can conclude that tetrahedral groups are completely ordered in the  $G_3$ -phase of RLM. TABLE 2 Atomic coordinates and isotropic thermal parameters of CsLiMoO<sub>4</sub> in the T<sup>2</sup><sub>d</sub> and C<sup>5</sup><sub>3</sub> phases The changes in the structure at G<sub>0</sub>-G<sub>1</sub> can be seen by comparison of the G<sub>0</sub>-coordinates (~300 K) taken from <sup>5</sup> (upper lines) to the G<sub>1</sub>-coordinates at 200 K (lower lines). The data for the ordered (I) and disordered (II) models with the corresponding B<sub>j</sub> values are presented. Both sets of the data are shown in the hexagonal unit cells (Z=3), Table 1.

	I. 0 r	dered m	odels	
	x	У	Z	Bj
Св	0 0	0 0	0 0	1,7(1)
₩	0 0	0 0	0,5 0,5021(1)	0,8(1)
0 <sub>1</sub>	<b>0</b> 0	0 0	0,3 <b>790</b> 0,3789(5)	7,2(1,7)
°2	-0,1613 -0,1772(8)	0,1613 0,1772(8)	0,5403 0,52 <b>7</b> 9(3)	6,3(1,5)
Li	0 0	0	0,250 0,251(2)	1,9(7)
	II. Disc	rdered	model	S
Ca	0 0	0 0	0 0	1,8(1)
W	0 0	0 0	0,5 0,5013(1)	1,0(1)
0 <sub>1</sub>	-0,0247 -0,0254(7)	0,0247 0,0254(7)	0 <b>,3777</b> 0,3809(3)	2,8(6)
°2	-0,1713 -0,1450(7)	0 <b>,1713</b> 0,1450(7)	0 <b>,5243</b> 0,55 <b>7</b> 9(4)	2,2(5)
°3	-0,1220 -0,1492(5)	-0,3179 -0,3165(6)	0,5 <b>49</b> 0 0,53 <b>35(6)</b>	2,2(5)
Li	0 0	0 0	0,250 0,251(2)	1,9(7)

Thus according to X-ray data it can be supposed that PT's in the group of crystals under consideration (or some of them) are the order-disorder type transitions. The spectroscopic data <sup>9</sup> have given the indirect confirmation of the supposition. For a definite choice of the PT's model the calorimetric study of the group of ferroelectrics was made.

#### 4. CALORIMETRIC DATA

Specific heat C<sub>p</sub> study of CLW <sup>10</sup> and CLM was carried out with the adiabatic calorimeter in temperature range 90-320 K. Discontinuous heating and cooling method was used. In order to determine the equilibrium temperature of transitions  $T_{c1}$ , the thermal hysteresis values  $\delta T_i$  and the entropy changes  $\delta S_i$  at  $T_{c1}$  the continuous heating and cooling experiments with the rate as low as  $dT/dt \sim 10^{-4}$  Ksec<sup>-1</sup> were made.

Temperature dependences of  $C_p$  for CLM and CLW are shown on Fig.1. The two  $C_p$ -anomalies correspond to  $G_0 \rightarrow G_1 \rightarrow G_2$  PT's. Both curves are similar, the  $C_p$  value at 300 K for CLW is 1,6 per cent greater than the corresponding one for CLM. For the determination of the excess part of heat capacity  $\Delta C_p$  associated with the first order transitions the graphic interpolation of the lattice heat capacity was used (dushed lines on Fig.1). The total entropy changes were estimated as  $\Delta S_i = \int \Delta C \cdot T^{-1} dT + \delta S_i$ .  $\delta S$ -values were calculated from the enthalpy changes  $\delta H_i$  at  $T=T_{ci}$  as  $\delta S_i = \delta H_i/T_i$ (Table 3).

Calorimetric data show that in cubic phases of both crystals the excess part of heat capacity exists in a broad region above  $T_{c1}$  ( $\sim T_{c1}$ +60 K), Fig.1. The increments to the entropy change  $\int \triangle CT^{-1} dT$  ( $T > T_{c1}$ ) are equal to  $\sim 0,05R$  (CLW) and 0,03R (CLM). These increments were not included to the  $\triangle S_1$  data of Table 3. The behaviour of the enthalpy at  $T=T_{c1}$  gives no indication to the "smearing" of PT's. It points out both at rather high quality of the crystals used in calorimetric experiments and at the fact that the  $\triangle C_p$  above  $T_{c1}$  is not connected with imperfections of the crystal. RLM single crystal



FIGURE 1 Temperature dependences of specific heat  $C_p(T)$  for the CLM (a) and CLW (b) crystals.

 $(\sim 0,09 \text{ g})$  was studied by DTA method. The sensitive copper-constantan thermocouple was used. The change of S at  $G_{\rightarrow}$  $G_1$  PT is equal to  $SS_1 =$ 0,22R. It was impossible to subdivide SS2 and  $\delta S_3$  for the close located PT's  $G_1 \rightarrow G_2$  and  $G_2 \rightarrow$  $\neq$  G<sub>3</sub>. The sum of the values ( $\simeq$  0,64R) is shown in Table 3. In order to check the reliability of the SS<sub>i</sub> data for RLM the same values were estimated by DTA for the CLW crystal. The comparison with the data  $\delta S_1$  and  $\delta S_2$  (Table 3) allows to conclude

that the error in  $\delta S_i$  for RLM does not exceed 10 per cent.

#### 5. MODEL THEORY

Both structural and calorimetric data have given some indications to the transitions of order-disorder type which connected with the ordering of the tetrahedral

## TABLE 3 Thermodynamical characteristics of phase transitions

The comparison between the experimental (E) and theoretical data is made. Two sets of the theoretical values correspond to molecular field (MF) and cluster approximation (CA) of the theory. R is the gase constant.

	CsLiMoO <sub>4</sub>			CsLiWO <sub>4</sub>		RbLiMoO4		
	E	MF	CA	E	MF	CA	E	MP
<sup>T</sup> c1, <sup>K</sup>	220	220	220	209	209	209	390	390
∆S <sub>1</sub> /R	0,3	0,71	0,4	0,34	0,71	0,4	-	0,71
δs <sub>1</sub> /R	0,17	0,55	0,32	0,18	0,55	0,32	0,22	0,55
T <sub>c2</sub> ,K	185	185	<b>18</b> 5	171	171	171	380	380
∆S <sub>2</sub> /R	0,33	1,77	1,2	0,28	1,77	1,2	-	1,74
δs <sub>2</sub> /R	0,12	0,77	0,3	0,16	0,77	0,3		0,74
T <sub>c3</sub> ,K	-	-	-	-	-	-	375	
δs <sub>3</sub> /r							0,64	0,03

group orientations in the structure with the temperature lowering. The following points were taken into consideration for the choice of the PT's model. In trigonal  $G_1$ -phases the WO<sub>4</sub> (MoO<sub>4</sub>) tetrahedra are disordered on three states around the 3-fold axis. In the monoclinic  $G_2$ -phases one of these three states is realized. In accordance to the symmetry we can conclude that in the cubic phase each tetrahedron has twelve equivalent states. Then the Hamiltonian of the model is

$$H = -\frac{1}{2} \sum V_{ij}(R-R')c_i(R) \cdot c_j(R)$$
(2)

where

$$c_{i}(R) = \begin{cases} 1 & \text{if a tetrahedron occupies the} \\ & \text{state i} \\ 0 & \text{in opposite cases} \end{cases}$$

12x12 Matrix of the nearest neighbours interaction  $V_{ij}$ in (2) has six independent constants. Some eigenvalues of the matrix can be found from the solution of the second-order equation. Even in mean field (MF) approximation the calculation of the free energy is difficult. The problem was simplified when it was supposed that some of  $V_{ij}$  are equal each other. The  $V_{ij}$  matrix in this case (4 independent values) has the block-diagonal form

$$\mathbf{V}_{ij} = \begin{array}{ccc} \mathbf{A} & \mathbf{C} & \mathbf{C} \\ \mathbf{V}_{ij} = \begin{array}{ccc} \mathbf{C} & \mathbf{A} & \mathbf{C} \\ \mathbf{C} & \mathbf{C} & \mathbf{A} \end{array} \quad \text{with } \mathbf{A} = \begin{array}{c} \mathbf{a} & \mathbf{b} & \mathbf{b} & \mathbf{b} \\ \mathbf{b} & \mathbf{a} & \mathbf{b} & \mathbf{b} \end{array}, \begin{array}{c} \mathbf{C} = \begin{array}{c} \mathbf{d} & \mathbf{d} & \mathbf{d} \\ \mathbf{d} & \mathbf{c} & \mathbf{d} & \mathbf{d} \\ \mathbf{d} & \mathbf{c} & \mathbf{d} & \mathbf{d} \\ \mathbf{b} & \mathbf{b} & \mathbf{b} & \mathbf{b} \end{array}$$

Free energy (F) of the simplified model can be found in the analytical form which was written in 7. It contains three order parameters  $\eta_{i}$  which were found from the conditions  $\partial F/\partial \eta_{i} = 0$ .

The free energy describing the PT's sequence in RLM crystal

$$T_d^2(Z=1) \rightarrow C_{3v}^5(Z=1) \rightarrow C_{2v}^2(Z=2) \rightarrow C_s^4(Z=2)$$
 (4)

is much more complicated and is not presented here.

The calculated MF temperature dependences of for CLM crystal are shown in Fig.2 in the comparison with some experimental data (see Figure Caption). MFvalues of  $\Delta S_i$  and  $\delta S_i$  for the three crystals are presented in Table 3. It is seen that the entropy changes are in poor agreement with the experimental data.



FIGURE 2 Order parameters behaviour versus relative temperature around  $T_{c1}$  and  $T_{c2}$  for the CLM crystal. The comparison of the experimental and theoretical data of mean field (dushed lines) and cluster approximations (full lines) is made. Full and empty circles correspond to the  $P_{s}(T)$  data 1 and to the square root of the quadrupole splitting ( $\Delta N$ ) of the 131Cs NMR lines 9, respectively. Both sets of the data were tied to the theoretical curve at  $T_{c29}$  In G<sub>2</sub>-phase the comparison with the 7Li NMR data 9 is made: triangles and crosses correspond to  $\Delta N$  (T) and  $[\Delta N(T)]^{1/2}$ , respectively.

The existence of  $\triangle C_p$  on a broad region above  $T_{c1}$ and the rather small experimental values of the transition entropies (see Fig.1,3 and Table 3) were supposed to be indications to the presence of strong correlations in the ordering system. Such correlations may be connected with the energetically unfavourable orientations of the neighbouring  $\mathrm{BO}_4$ -tetrahedra. Each  $\mathrm{MoO}_4$  (WO<sub>4</sub>) tetrahedron in the G<sub>0</sub>-structure is surrounded by four LiO<sub>4</sub> tetrahedra and vice versa. The structural data have shown that in most cases the PT's in crystals with tetrahedral groups are connected with the tilts of tetrahedra without essential distortions <sup>11</sup>. It indicates on the high energy of such configurations of the BO<sub>4</sub>-groups which lead to distortions of LiO<sub>4</sub>-tetrahedron in cibic phase.

This correlations were taken into consideration in the cluster approximation (CA) of the model theory. Group of four  $BO_{4}$  tetrahedra was choosen as a cluster. Enegies of configurations which distort the LiO<sub>4</sub> tetrahedron have been equated to infinity. Then from  $12^4 =$ 20736 configurations only 8746 ones give contributions to the free energy. The analytical form



FIGURE 3 Excess part of specific heat  $\triangle C_p$  versus relative temperature for CLM. Full lines - cluster approximation of the theory, dotted lines - experimental data extracted from the  $C_p(T)$  dependence shown on Fig.1.

of F is very complicated and will not be written here. The calculated values of  $\gamma_i(T)$  and  $\Delta C_p(T)$  for the CLM crystal are shown on Fig.2 and 3. The CA-values of  $\Delta S_i$  and  $\delta S_i$  are included in Table 3.

It is clearly seen that in the cluster approximation the agreement of theoretical results with the experimental values is more satisfactory than that of the MF-data. But there exist some definite differences in  $\triangle C_p(T)$  above  $T_{c1}$  and  $\triangle S_i$ ,  $\delta S_i$  values.

#### 6. SUMMARY

In this paper the results of theoretical and experimental study of PT's in ferroelectrics of CsLiMoO<sub>4</sub> type were presented. The following conclusions could be made on the basis of the study.

(i) PT's in the family of ferroelectrics belong to the order-disorder type, and the ordering process of tetrahedral groups determines the mechanism of PT's.

(ii) Statistical model of PT's was treated and analyzed in the MF- and CA approximations. The last approximation, where some correlations of the tetrahedral tilts in the G<sub>o</sub>-phase were taken into account, the correspondence between the experimental and theoretical data was considerably improved.

(iii) The model gives the correct qualitative description of the symmetry changes and the thermodynamical behaviour of the crystals in a broad temperature range around the PT's. But the full quantitative description has not been obtained yet. Some new experimental study on the properties of the  $G_o$ -phase should be made in order to select more satisfactory model of the correlations.

#### REFERENCES

- K.S.Aleksandrov, A.T.Anistratov, S.V.Melnikova, P.V.Klevtsov, A.I.Kruglik, V.N.Voronov, <u>Phys.Stat</u>. <u>Solidi (a) 67</u>, 377 (1981); <u>Ferroelectrics 36</u>, 399 (1981).
- 2. K.S.Aleksandrov, A.T.Anistratov, S.V.Melnikova, A.I.Kruglik, L.I.Zherebtsova, L.A.Shabanova, <u>Fizika Tverdogo Tela 24</u>, 1094 (1982).

- S.V.Melnikova, P.V.Klevtsov, L.I.Zherebtsova, A.I. Kruglik, L.A.Shabanova, V.N.Voronov, <u>Fizika Tverdogo Tela 24</u>, 2862 (1982).
- 4. R.F.Klevtsova, P.V.Klevtsov, K.S.Aleksandrov, Doklady Akad.Nauk USSR, 255, 1379 (1980).
- 5. K.Okada, J.Ossaka, <u>Acta Cryst.B36</u>, <u>657</u> (1980).
- 6. K.Okada, J.Ossaka, <u>J.Solid State Chem.</u>, <u>37</u>, 325 (1981).
- 7. K.S.Aleksandrov, D.H.Blat, V.I.Zinenko, I.M.Iskornev, A.I.Kruglik, I.N.Flerov, <u>Ferroelectrics</u>, <u>54</u>, 233 (1984).
- 8. A.I.Kruglik, R.F.Klevtsova, K.S.Aleksandrov, <u>Doklady Akad.Nauk USSR</u>, <u>271</u>, 1388 (1983).
- 9. K.S.Aleksandrov, Yu.N.Ivanov, A.D.Schäfer, <u>Ferro-</u> <u>electrics</u>, <u>55</u>, 3 (1984).
- 10. I.N.Flerov, I.M.Iskornev, M.V.Gorev, <u>Ferroelect-</u> <u>rics Letters</u>, <u>44</u>, 235 (1983).
- K.S.Aleksandrov, A.I.Kruglik, V.I.Zinenko, <u>Ferro-electrics</u>, <u>52</u>, 157 (1984).