EISEVIER

Contents lists available at ScienceDirect

Journal of Magnetism and Magnetic Materials

journal homepage: www.elsevier.com/locate/jmmm



DFT investigation of electronic structures and magnetic properties of halides family MeHal₃ (Me=Ti, Mo,Zr,Nb, Ru, Hal=Cl,Br,I) one dimensional structures



A.A. Kuzubov a,b , E.A. Kovalev a,b , M.I. Popov a , A.S. Kholtobin a , N.S. Mikhalev a,b , M.A. Visotin a,b , A.S. Fedorov a,b,*

- ^a Siberian Federal University, av. Svobodny 79, Krasnoyarsk 660041, Russia
- ^b Kirensky Institute of Physics, Akademgorodok 50, Krasnoyarsk 660036, Russia

ARTICLE INFO

Keywords: Transition metal trihalides Magnetic properties Thermoelectric conversion Density functional theory

ABSTRACT

Using DFT GGA calculations, electronic structure and magnetic properties of wide family of transition metal trihalides (TMHal $_3$) (Zr, Ti and Nb iodides, Mo, Ru, Ti and Zr bromides and Ti or Zr chlorides) are investigated. These structures consist of transition metal atoms chains surrounded by halides atoms. Chains are connected to each other by weak interactions. All TMHal $_3$ compounds were found to be conductive along chain axis except of MoBr $_3$ which is indirect gap semiconductor. It was shown that NbI $_3$ and MoBr $_3$ have large magnetic moments on metal atoms (1.17 and 1.81 μ_B , respectively) but other TMHal $_3$ materials have small or zero magnetic moments. For all structures ferromagnetic and anti-ferromagnetic phases have almost the same energies. The causes of these properties are debated.

1. Introduction

At present, as new nanostructure synthesis methods are developed, more and more attention is paid for synthesis and description of new one-dimensional structures properties. In practice, such nanostructure chains are often linked together by weak dispersive interactions. Thus, they may serve as a one-dimensional conductors or semiconductors that should lead to a strong anisotropy of their optical properties. Also such materials having semiconductor properties can be used in practice to create different optical elements: IR sensors, absorbing devices, etc. Weak interactions between neighboring chains due to van der Waals forces should lead to softening of the phonon vibrations along the directions perpendicular to the chain axis and, consequently, to a decrease in the thermal conductivity λ along these directions. This can also be used in practice to create new thermoelectric converters (TC) [1], where the efficiency is defined by thermoelectric figure-of-merit, ZT [2], as a product of working temperature T, Seebeck coefficient squared S^2 , electrical conductivity σ , divided by the thermal conductivity. Therefore "phonon engineering", i.e creation of materials having low thermal conductivity due to structural fragment separation at the nanoscale is now represented as one of the most promising ways to create highly effective TC. Along with that, TC material is demanded to have large density of states near the Fermi level. This allows one to suppose that one-dimensional transition metal trihalides (TMHal $_3$) having weak van-der-Waals interaction between chains would show low heat conductance in the directions perpendicular to the chain axis and high density of states in the vicinity of Fermi level due to presence of metal atoms having d(f) electrons. It is assumed that these materials can serve as effective materials for TC construction.

Even though the family of TMHal₃ has been known for more than 50 years [3,4], and their structure is well-investigated, there is only a few data available regarding physical properties of these compounds, particularly, electronic structure and magnetic properties [5]. Purpose of this work is to fill this gap. TMHal₃ family compounds can adopt various structures consisting of octahedra bonded to each other through the vertexes and forming chains in particular direction (ReO₃, RhF₃, AlCl₃, TiI₃-type structures). Amongst them, the TiI₃-type is somewhat outstanding in terms of having quasi-one-dimensional structure of these chains (see Fig. 1). These unique compounds can be, in turn, distinguished into hexagonal TiI₃-type and orthorhombic RuBr₃-type.

It is known that temperature increasing up to 323 K may cause structural transition from hexagonal to orthorhombic crystal system in ${\rm TiI}_3$ [6]. Semiconducting nature of some ${\rm TiI}_3$ -like structures was reported by Angelkort et al. [7]. In this paper, using quantum chemical calculations we examine the electronic structure and magnetic proper-

^{*} Corresponding author at: Kirensky Institute of Physics, Akademgorodok 50, Krasnoyarsk 660036, Russia. E-mail address: alex99@iph.krasn.ru (A.S. Fedorov).

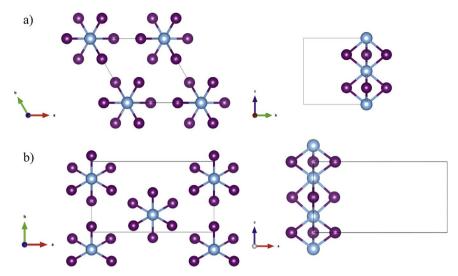


Fig. 1. Unit cells for transition metal trihalides. a) hexagonal, b) orthorhombic structure.

ties of a wide range of $TMHal_3$ compounds adopting quasi-one-dimensional structure, namely, Zr, Ti and Nb iodides, Mo, Ru, Ti and Zr bromides and Ti or Zr chlorides.

2. The calculations details

All quantum chemical calculations were performed within the framework of density functional theory using plane wave basis set and projector augmented wave method [8,9] as implemented in Vienna Ab-initio Simulation Package [10–13]. GGA-PBE spin polarized exchange-correlation functional [14] and Grimme D3 [15] correction for van-der-Waals interactions were used for electronic and structural optimization in all cases. Residual forces acting on atoms being less than $10^{-3}\,\mathrm{eV/Å}$ were used as stopping criteria for cell vectors and geometry optimization. Monkhorst-Pack k-point first Brilloin zone sampling [16] was used with k-point mesh containing $6\times6\times6$ points along three translation vectors.

3. Results

First, unit cells of all mentioned above transition metal trihalides were optimized. Table 1S of Supplementary material summarizes data regarding unit cell parameters, space group and interatomic distances for the structures under investigation. All of these compounds adopt "chain" structure with chemically bonded atoms forming columns along the ${\bf c}$ translation vector.

Fig. 1 shows two different crystal structures being realized in this type of compounds. Zirconium trihalides and ruthenium bromide were found to be diamagnetic while titanium, niobium and molybdenum compounds possess magnetic moment.

It was established that TM atoms antiferromagnetic ordering were more favorable than ferromagnetic one for all cases, see Table 1.

Analysis of spin density shows it is mainly located on TM atoms, and, depending on the nature of metal, unpaired electrons occupy different orbitals, see Fig. 2. Small energy differences between ferroand antiferromagnetic states (maximum value of 0.05 eV corresponds to MoBr₃) witness the weak exchange between TM d-orbitals where unpaired electrons are located.

Band structure and density of state (DOS) calculations showed (see Fig. 3, Fig. 1S of Supplementary material) metallic conductivity in all compounds except of $MoBr_3$ which is indirect gap semiconductor with a gap of ~1.1 eV (Fig. 3c). The most valuable contribution to the DOS nearby the Fermi level rises from TM d-electrons, see Fig. 3 and Fig. 1S.

Once TMH_3 compounds adopt "chain" structure along the c vector,

Table 1Magnetic moment (M) on the TM atoms and energy difference $\Delta E = E_{\text{ferro}} - E_{\text{antiferro}}$ between ferro- and antiferromagnetic states.

| Compound | $M,\mu_{ m B}$ | ΔE , eV |
|------------------------|----------------|-----------------|
| ZrI ₃ | 0.000 | _ |
| TiI ₃ (hex) | 0.249 | 0.005 |
| TiI_3 | 0.300 | 0.004 |
| NbI_3 | 1.166 | 0.018 |
| MoBr ₃ | 1.805 | 0.053 |
| RuBr ₃ | 0.00 | _ |
| TiBr ₃ | 0.129 | 0.001 |
| ZrBr ₃ | 0.000 | |
| TiCl ₃ | 0.043 | 0.002 |
| ZrCl ₃ | 0.000 | _ |

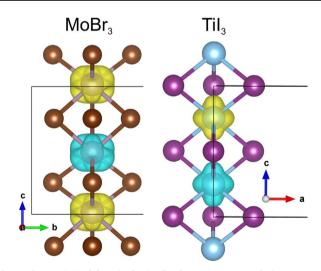


Fig. 2. Characteristic of the spin density distribution in $MoBr_2$ and TiI_3 compounds. Yellow and blue areas correspond to spin -up and spin-down densities, respectively. (For interpretation of the references to color in this figure, the reader is referred to the web version of this article.)

the issue of nature of bonding between chains is fundamental. The analysis of TMH $_3$ band structures along the directions other than $\Gamma\text{--}Z$ (corresponding to the chain axis) showed band dispersion suggesting chemical bonding between neighboring chains, in agreement with earlier tight-binding calculations [5]. Band structures of TiI $_3$ (orth), TiI $_3$ (hex), TiBr $_3$, ZrBr $_3$, TiCl $_3$, ZrCl $_3$ nearby the Fermi level is analogous to that of ZrI $_3$. Its main feature is significant band dispersion along $\Gamma\text{--}Z$ direction and weak dispersion along perpendicular directions. Thus, a

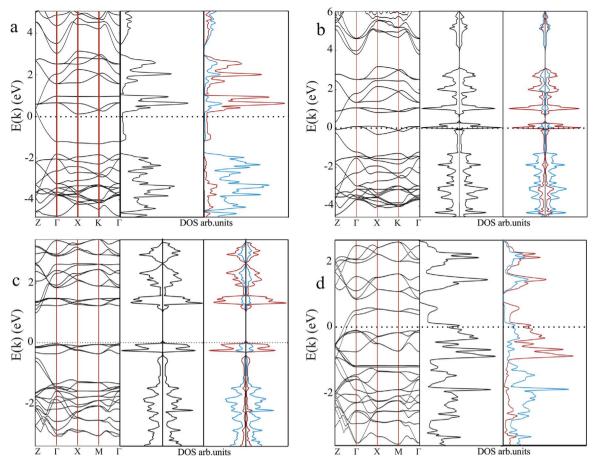
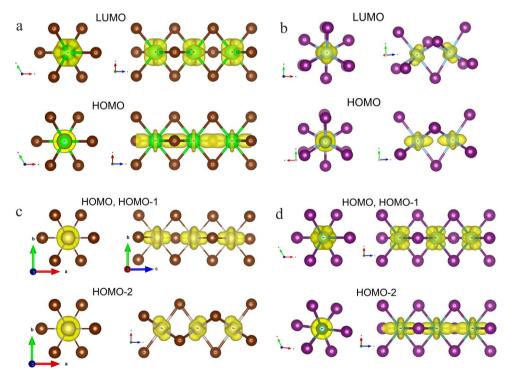


Fig. 3. Band structure, total (DOS) and partial densities of states (POS) for antiferromagnetic ground state of (a) ZrI_3 , (b) NbI_3 ; (c) $MoBr_3$, (d) $RuBr_3$. Red and blue lines correspond to the PDOS of metal and halogen atoms, respectively. (For interpretation of the references to color in this figure, the reader is referred to the web version of this article.)



 $\textbf{Fig. 4.} \ \ \textbf{Electron density distribution for bands in the Fermi level vicinity for (a) } \ \ \textbf{ZrI}_3; \ (b) \ \ \textbf{TiI}_3; \ (c) \ \ \textbf{RuBr}_3; \ (d) \ \ \textbf{NbI}_3.$

hole pocket will be formed along the Γ -Z direction causing hole type of conductivity in these systems. In order to elucidate its nature, spatial electron density distributions for the corresponding band was analyzed. According to the Fig. 4, the band is formed by the interacting transition metal d_z^2 -orbitals. This fact confirms the suggestion regarding weak exchange interaction formation through the direct exchange between transition metal d-orbitals. Overlapping area depends on the TM-TM distance and the size of d-orbitals which is changed according to the TM atom size belonging either to third or fourth rows. This, in turn, influences to the interaction energy. So, the size of the d-orbital plays a key role in the magnetic moments formation, d₂²-orbitals of 3d Ti atoms are too small to overlap with each other which leads to the localization of d-electrons giving rise to magnetic moment whereas 4d elements have large orbitals with larger overlapping resulting in zero magnetic moment. Fermi level crossing bands of RuBr₃ are similar to ones described above but in directions other than $Z-\Gamma$ these states are vacant witnessing electron type of conductivity. Bands formed by the d_z²-orbitals of TM atoms are lower in energy for NbI₃ and MoBr₃ (see Figs. 3b and 3c), thus being fully occupied and not contributing to magnetic moment of the system, see Fig. 4d.

Weak band dispersion in the proximity of Fermi level in all directions is characteristic for NbI₃ and MoBr₃. Spatial electron density distribution analysis showed that TM half-occupied d-orbitals don't interact with each other being located distant from TM-Hal bonds, see Fig. 4d. This results in the localization of spin density (Fig. 2) and magnetic moment increasing (see Table 1).

4. Conclusions

By DFT quantum chemical calculations the electronic structure and magnetic properties of the wide family of transition metal trihalides having chain structures are investigated. It is shown the origin of the large metal atom magnetic moments inside NbI₃ and MoBr₃ is strong localization of the metal d-orbitals, flat band structure and large density of states near the Fermi level. In other cases there is a substantial dispersion of electronic bands along the chain axis at the Fermi level which is the cause of conduction along the chain axis. In all cases the ferromagnetic and antiferromagnetic phases have almost the same energies because of weak overlapping of d-orbitals of the nearest transition metal atoms. It is assumed that TiI₃, TiCl₃ and NbI₃ materials due to the high density of states near the Fermi level and weak interaction between chains can be promising materials for thermoelectric applications.

Acknowledgments

This work was supported by the Russian Science Foundation, Project no. 16-13-00060. N.S.M. thanks the Russian Foundation for Basic Research, project RFBR 16-32-60003 mol_a_dk, for the finan-

cial support of ZrI₃, TiI₃ and NbI₃ electronic structure calculations.

Appendix A. Supplementary material

Supplementary data associated with this article can be found in the online version at doi:10.1016/j.jmmm.2016.12.054.

References

- D.M. Rowe, Thermoelectric waste heat recovery as a renewable energy source / D.M. Rowe, Int. J. Innov. Energy Syst. Power 1 (1) (2006) 13–23.
- [2] Y. Faraji, et al. Electric Power Generation using Thermoelectric Cells and Evacuated Tube Solar Collectors/Y. Faraji, A. Akbarzadeh, J. Leblanc, R. Singh, C. Bourke// Solar 2010, in: Proceedings of the 48th AuSES Annual Conference —— P. 1–13, 2010
- W.A. Baker, et al., Magnetic properties of zirconium (III) and hafnium (III) iodides / W.A. Baker, A.R. Janus, J. Inorg. Nucl. Chem. 2 (12) (1964) 2087–2097. http://dx.doi.org/10.1016/0022-1902(64)80153-6.
- [4] C. Michel, et al., Structural relationships in compounds with R3c symmetry / C. Michel, J. M. Moreau, W.J. James, Acta Crystallogr. Sect. B B27 (1971) 501. http://dx.doi.org/10.1107/S0567740871002437.
- [5] J. Lin, et al., Dimensional diversity in transition metal trihalides/J. Lin, G. J. Gordon, lnorg. Chem. 32 (1993) 1476. http://dx.doi.org/10.1021/ic00060a025
- Gordon, Inorg. Chem. 32 (1993) 1476. http://dx.doi.org/10.1021/ic00060a025. [6] H. Hillebrecht, et al., About trihalides with Til₃ chain structure: proof of pair forming of cations in β-RuCl3 and RuBr3 by temperature dependent single crystal X-ray analyses/H. Hillebrecht, T. Ludwig, G. Thiele, Z. für Anorg. und Allg. Chem. 630 (13–14) (2004) 2199–2204. http://dx.doi.org/10.1002/zaac.200400106.
- [7] J. Angelkort, et al., Temperature-dependent neutron diffraction on TiI₃/J. Angelkort, A. Senyshyn, A. Schönleber, S. van Smaalen, Z. für Krist. 226 (8) (2011) 640–645. http://dx.doi.org/10.1524/zkri.2011.1370.
- [8] P.E. Blöchl, Projector augmented-wave method/P.E. Blöchl, Phys. Rev. B. 50 (24) (1994) 17953-17979. http://dx.doi.org/10.1103/PhysRevB.50.17953.
- [9] G. Kresse, et al., From ultrasoft pseudopotentials to the projector augmented-wave method/G. Kresse, D. Joubert, Phys. Rev. B 59 (3) (1999) 1758–1775. http:// dx.doi.org/10.1103/PhysRevB.59.1758.
- [10] G. Kresse, et al., Efficient iterative schemes for ab initio total-energy calculations using a plane-wave basis set/G. Kresse, J. Furthmüller, Phys. Rev. B. 54 (16) (1996) 11169–11186. http://dx.doi.org/10.1103/PhysRevB.54.11169.
- [11] G. Kresse, et al., Ab initio molecular dynamics for liquid metals/G. Kresse, J. Hafner, Phys. Rev. B. 47 (№ 1) (1993) 558–561. http://dx.doi.org/10.1103/ PhysRevB.47.558.
- [12] G. Kresse, et al., Ab initio molecular-dynamics simulation of the liquid-metal-amorphous-semiconductor transition in germanium/G. Kress, J. Hafnere, Phys. Rev. B. 49 (20) (1994) 14251–14269. http://dx.doi.org/10.1103/PhysRevB.49.14251.
- [13] G. Kresse, et al., Efficiency of ab-initio total energy calculations for metals and semiconductors using a plane-wave basis set / G. Kresse, J. Furthmüller, Comput. Mater. Sci. 6 (1) (1996) 15–50. http://dx.doi.org/10.1016/0927-0256(96)00008-
- [14] J.P. Perdew, et al., Generalized gradient approximation made simple / J.P. Perdew, K. Burke, M. Ernzerhof, Phys. Rev. Lett. 77 (18) (1996) 3865–3868. http:// dx.doi.org/10.1103/PhysRevLett.77.3865.
- [15] S. Grimme, et al., A consistent and accurate ab initio parametrization of density functional dispersion correction (DFT-D) for the 94 elements H-Pu/S. Grimme, J. Antony, S. Ehrlich, H. Krieg, J. Chem. Phys. 132 (15) (2010) 154104. http:// dx.doi.org/10.1063/1.3382344.
- [16] H.J. Monkhorst, et al., Special points for Brillouin-zone integrations/H.J. Monkhorst, J.D. Pack, Phys. Rev. B. 13 (12) (1976) 5188–5192. http://dx.doi.org/ 10.1103/PhysRevB.13.5188.