



Revisiting the BaBiO₃ semiconductor photocatalyst: synthesis, characterization, electronic structure, and photocatalytic activity

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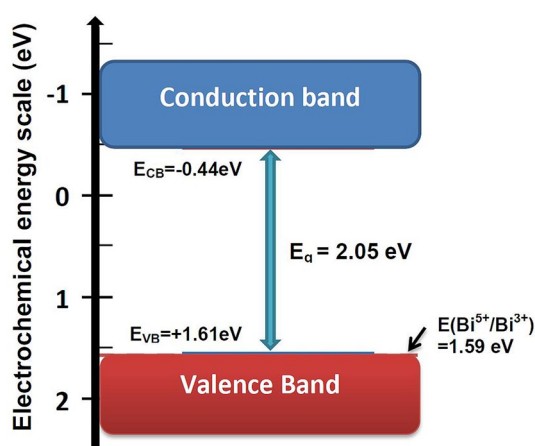
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Abstract

This article revisits the properties of BaBiO₃ examined extensively in the last two decades because of its electronic properties as a superconductor and as a semiconductor photocatalyst. Solid-state syntheses of this bismuthate have often involved BaCO₃ as the barium source, which may lead to the formation of BaBiO₃/BaCO₃ heterostructures that could have an impact on the electronic properties and, more importantly, on the photocatalytic activity of this bismuthate. Accordingly, we synthesized BaBiO₃ by a solid-state route to avoid the use of a carbonate; it was characterized by XRD, SEM, and EDX, while elemental mapping characterized the composition and the morphology of the crystalline BaBiO₃ and its thin films with respect to structure, optoelectronic, and photocatalytic properties. XPS, periodic DFT calculations, and electrochemical impedance spectroscopy ascertained the electronic and electrical properties, while Raman and DRS spectroscopies assessed the relevant optical properties. The photocatalytic activity was determined via the degradation of phenol in aqueous media. Although some results accorded with earlier studies, the newer electronic structural data on this bismuthate, together with the photocatalytic experiments carried out in the presence of selective radical trapping agents, led to elucidating some of the mechanistic details of the photocatalytic processes that previous views of the BaBiO₃ band structure failed to address or clarify. Analytical refinement of the XRD data inferred the as-synthesized BaBiO₃ adopted the C_{2/m} symmetry rather than the I_{2/m} structure reported earlier, while Tauc plots from DRS spectra yielded a bandgap of 2.05 eV versus the range of 1.1–2.25 eV reported by others; the corresponding flatband potentials were 1.61 eV (E_{VB}) and –0.44 eV (E_{CB}). The photocatalytic activity of BaBiO₃ was somewhat greater than that of the well-known Evonik P25 TiO₂ photocatalyst under comparable experimental conditions.

Graphic abstract



Keywords Barium bismuthate · Visible-light-active photocatalyst · Photocatalytic activity · Bandgaps · Flatband potentials

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1 Introduction

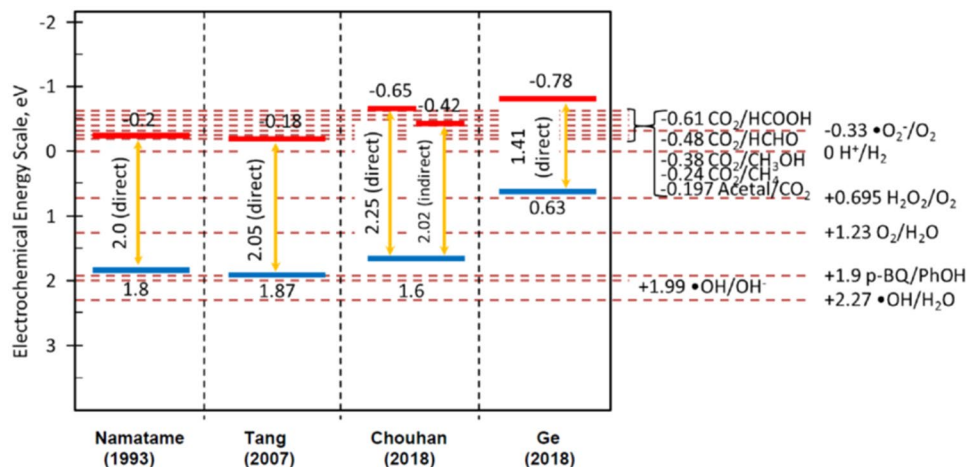
The past two decades have witnessed a significant renaissance of the bismuthate BaBiO_3 from both a technological and scientific viewpoint [1–11], especially as a superconductor [1, 3, 5–7] because of its electronic properties, and as a promising photocatalyst toward the degradation of organic pollutants [2], water splitting [9, 10], CO_2 reduction [4], and as a prospective photovoltaic material [11]. The unique properties of BaBiO_3 are largely the result of the occurrence of both Bi^{3+} and Bi^{5+} oxidation states that have tended to complicate studies of this bismuthate. For instance, Namatame et al. [1] found the peak-to-peak splitting of the Bi_{6s} band to be large, while the minimum gap was much smaller, consistent with the large direct optical gap of ~ 2 eV and a smaller indirect transport gap of ~ 0.5 eV. Their comparison of the photoemission, optical absorption, and transport data on BaBiO_3 suggested the existence of polaron energy levels within the bandgap of BaBiO_3 that accommodate thermally excited charge carriers. Kumar and coworkers [5] synthesized this bismuthate from Bi_2O_3 and BaCO_3 via a two-step solid-phase heating method that, when sintered in oxygen, yielded a single-phase BaBiO_3 with monoclinic $I_{2/mnm}$ symmetry from high-resolution XRD, while XPS confirmed the presence of two valence states of bismuth: Bi^{3+} and Bi^{5+} . Optical spectroscopy revealed a direct bandgap of ~ 2.2 eV and a possible indirect bandgap of ~ 0.9 eV, which when combined with the activation energy for conduction of 0.25 eV (*ac* impedance spectroscopy) led Kumar et al. [5] to infer a polaron-mediated conduction mechanism to prevail in BaBiO_3 . Importantly, the optical absorption spectra given by these authors revealed yet another linear section in the Tauc plot that yields an optical bandgap of 1.67 eV (see Fig. S1 in Supplemental Information). For their part, using a two-step heating solid-state protocol, Chouhan

et al. [11] prepared BaBiO_3 from Bi_2O_3 and BaCO_3 for photovoltaic applications. Their analysis of the optical transition spectra of thin films of BaBiO_3 featured both a direct ($E_g = 2.25$ eV) and an indirect ($E_g = 2.02$ eV) bandgap, while UPS revealed a flatband potential E_{VB} of 1.6 eV [11].

The photocatalytic properties of BaBiO_3 prepared from $\text{Ba}(\text{NO}_3)_2$ and $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ in aqueous media were first reported by Tang et al. [2] for the mineralization of acetaldehyde and methylene blue. A Tauc plot from diffuse reflectance spectra yielded a direct bandgap E_{bg} of 2.05 eV; the flatband potentials E_{VB} and E_{CB} estimated theoretically by the Butler-Ginley method gave 1.87 eV and -0.18 eV, respectively (see also Fig. S2). The authors attributed the photocatalytic activity of the BaBiO_3 particulates to the formation of H_2O_2 on the particle surface; the reduction half-reaction was left silent. Similarly, Ge et al. [9] prepared BaBiO_3 from barium and bismuth acetates dissolved in a mixture of acetic acid, 2-methoxyethanol, and acetylacetone with their goal of applying the resulting bismuthate system in the photocatalytic water splitting process. Optical transmission spectra of BaBiO_3 thin films yielded $E_{\text{bg}} = 1.41$ eV for direct optical transitions [9], a bandgap that differed significantly from those reported by others. The Mott–Schottky technique gave a flatband band potential E_{CB} of ca. -0.79 eV, approximately 0.5 eV greater than values determined elsewhere [2]. By contrast, Khraisheh et al. [4] examined the photocatalytic reduction of CO_2 to methane and CO under visible light in the presence of BaBiO_3 prepared using a high-temperature solid-phase synthesis (1100 °C for 4–5 h) using Bi_2O_3 and BaCO_3 as precursors; absorption spectroscopy revealed the absorption band edge to be located at 2.04 eV ($=E_{\text{bg}}$).

A compilation of the band structures of BaBiO_3 based on earlier analyses [1, 2, 9, 11] is illustrated in Fig. 1, which shows some of the disparities in regard to both bandgaps and VB/CB flatband potentials. Although the figure shows

Fig. 1 Band structures of BaBiO_3 derived from literature analyses [1, 2, 9, 11]. The dashed lines denote various half-reactions and their corresponding redox potentials



only four examples of direct/indirect bandgaps for this bismuthate, other experimental determinations of the bandgap energy have revealed far greater disparities. For instance, Uchida et al. [12] found a direct E_{bg} of 2.0 eV for BaBiO_3 , while others described indirect bandgaps E_{bg} of 0.2 eV [13], 0.5 eV [14], 0.8 eV [15], 1.1 eV [16], and 1.94–2.02 eV depending on temperature and preparative mode [10]. Theoretical computations of direct and indirect bandgap energies fare no better, as they are strongly dependent on the assumptions and DFT approaches taken. For instance, computed direct bandgaps vary wildly from 1.1 to 3.00 eV, whereas the indirect bandgap energies of this bismuthate range from 0.0 to 1.63 eV [17]. Also listed in Fig. 1 are some relevant redox potentials for the reduction of CO_2 (and others) to such reduced products as formaldehyde, methanol and methane. Clearly, information on the electronic structure of BaBiO_3 remains rather partial and controversial.

An important issue addressed in this study is the very nature of the BaBiO_3 obtained from various synthetic routes. Most researchers [4, 5, 11] used either BaCO_3 or some carbon-containing precursor to synthesize BaBiO_3 for which the presence of carbonate species will affect the photocatalytic activity as recently demonstrated by us [18, 19]; for example, even trace amounts of SrCO_3 enhanced the photocatalytic activity of strontium bismuthates. Another point addressed is the application of the Butler-Ginley method to determine the flatband potentials E_{VB} and E_{CB} as the method fails to produce accurate flatband potentials as demonstrated from a comparison of experimental and theoretical data [20, 21]. More importantly, within the context of the present study, earlier studies devoted to BaBiO_3 failed to address adequately the mechanistic details of the photocatalytic activity of this important bismuthate.

Accordingly, the principal objectives in revisiting the characteristics of barium bismuthate in the present study aimed at (i) a synthesis of BaBiO_3 that is carbonate-free, (ii) a detailed characterization of the bismuthate's band structure; and (iii) at accessing some of the mechanistic stages of the photocatalytic activity using phenol as the model substrate and various selected radical scavengers.

2 Experimental techniques and methods

2.1 Materials and synthesis

All chemical reagents for the solid-state synthesis of BaBiO_3 were of analytical grade (> 99.5% purity; ACROS Chemicals) and used without further purification. Unless noted otherwise, most experiments were performed using powdered BaBiO_3 samples prepared by a two-step heating solid-phase method using Bi_2O_3 and $\text{Ba}(\text{NO}_3)_2$ as precursors. Thus, a stoichiometric mixture of the latter two was prepared and

thoroughly ground, followed by annealing at 650 °C for 10 h (first stage) that led to the complete degradation of $\text{Ba}(\text{NO}_3)_2$ to barium and gaseous nitrogen oxides. In the second stage, the resulting mixture was ground and pressed again, a process repeated every 12 h during annealing at 730 °C over a period of 72 h.

Typically, studies of the optical properties of BaBiO_3 were conducted with thin films prepared by dissolving powdered BaBiO_3 in boiling glacial acetic acid, followed by placing the solution onto a quartz substrate using the dip-coating technique, after which it was annealed in a preheated muffle furnace for 72 h at 730 °C.

2.2 Characterization of the as-prepared BaBiO_3

Powder diffraction data of BaBiO_3 for the Rietveld refinement analysis by the TOPAS 4.2 system [22] were collected at room temperature on a Bruker D8 ADVANCE powder diffractometer (Cu-K α radiation) equipped with a linear VANTEC detector; the step size of 2θ was 0.016 degrees; counting time was 1.5 s/step. The morphology of BaBiO_3 was investigated by Scanning Electron Microscopy (SEM, TESCAN; accelerating voltage, 20 kV), while the elemental composition was established by Energy Dispersive X-ray Spectroscopy (EDX; model X-MaxN; Oxford Instruments). As well, the elemental composition of the surface of pure BaBiO_3 , together with the valence band potential were determined by X-ray Photoelectron Spectroscopy (XPS) using the Thermo Fisher Scientific Escalab 250Xi spectrometer (Al K α radiation, 1486.6 eV; spectral resolution, 0.5 eV); the C_{1s} carbon line (C–C bond) was the reference at the binding energy of 284.8 eV [23].

The specific surface area of the as-synthesized barium bismuthate particles, pretreated at 350 °C for 6 h, was obtained by the BET method using the Quadrasorb SI surface area analyzer (Quantachrome Instruments) and the curves of the physical adsorption–desorption of krypton Kr; the corresponding sizes of BaBiO_3 particles were measured using the Nanotrak Ultra analyzer (Anton Paar GmbH).

Raman spectra of pure BaBiO_3 were recorded at ambient temperature in the 80–1500 cm^{-1} spectral region using the Bruker SENTERRA Raman spectrometer (resolution, 2 cm^{-1} ; excitation laser wavelength, 785 nm; laser beam power, 1 mW). Other optical properties of BaBiO_3 were assessed using thin films deposited on a quartz substrate. Transmission spectra were recorded in the spectral range 190–800 nm under ambient conditions using a UV-1800 UV/vis spectrophotometer. The film's width was controlled by means of optical microscopy using a Zeiss–Axio Imager. A 2 m optical microscope (Carl Zeiss Microscopy Deutschland GmbH, Germany).

The electrophysical properties of as-synthesized BaBiO_3 samples were ascertained by electrochemical impedance

spectroscopy (EIS). The NOVOCONTROL BDS dielectric spectrometer (Novocontrol Technologies GmbH & Co. KG, Germany) provided precision measurements of the complex conductivity over a wide frequency range from 0.01 to 40 MHz and temperatures from -150 to 275 °C. The powdered samples were pressed into self-supported pellets (diameter 8.9 mm; thickness 0.62 mm).

2.3 Photocatalytic activity

The oxidative degradation of phenol in aqueous media, carried out under otherwise identical conditions used earlier for strontium bismuthates [24], was used to assess the photocatalytic activity of BaBiO₃ using a batch-type reactor with a lateral quartz window. The aqueous suspension of BaBiO₃ (loading, 1.0 g L⁻¹; pH 7.0; volume, 300 mL) was pretreated in an ultrasonic processor for 10 min, followed by the addition of phenol (purity, 99.5%; Aldrich) for a total concentration of 100 ppm (100 mg L⁻¹ or 1.06 mmol L⁻¹). The suspension was subsequently stirred with a magnetic stirrer for 1 h in the dark to achieve adsorption–desorption equilibrium, after which it was irradiated with a 150-W xenon lamp (OSRAM) at wavelengths above 300 nm (cutoff filter; incident light irradiance, 7 mW cm⁻²). Aliquots were collected at 30-min intervals, filtered through a 0.2 μm Minisart filter to remove solid particles prior to the HPLC analysis of the time-dependent phenol concentration (1260 Infinity liquid chromatograph; UV–Vis detector; Agilent Technologies C18 column); the mobile phase was a 50/50 v/v mixture of methanol/water; the detection wavelength was 210 nm. For comparison, the photocatalytic activity of Evonik's P25 TiO₂ toward the degradation of phenol was also determined under otherwise similar conditions as for the BaBiO₃ samples.

The photocatalytic activity of the samples was estimated from the rates of the photodegradation of phenol using $C(t) = C_0 - kt$, where C_0 is the initial phenol concentration, t is the irradiation time, and k is the zero-order rate constant.

To unravel the mechanistic details of the photocatalytic degradation of phenol, reactions were also carried out in the presence of selective radical scavengers that interact either with the photogenerated electrons (e^-) or with holes (h^+) subsequent to absorption of light energy equal to or greater than the bismuthate's bandgap. These charge carriers lead to the formation of reactive oxygen species (ROS) on the surface of BaBiO₃ particulates: hydroxyl radicals $\{\text{OH}^- (\text{H}_2\text{O}) + h^+ \rightarrow \cdot\text{OH} (+\text{H}^+)\}$ and superoxide radical anions ($\text{O}_2 + e^- \rightarrow \text{O}_2^{\cdot-}$), together with hydroperoxyl radicals $\cdot\text{O}_2\text{H}$ [25]. The selected radical scavengers interact with these species and thus prevent them from participating in the photo-oxidative degradation of phenol. For instance, isopropyl alcohol (IPA; purity > 99.9%; Panreac) scavenges $\cdot\text{OH}$ radicals, while *p*-benzoquinone (*p*-BQ, purity > 98%, Sigma-Aldrich) traps $\text{O}_2^{\cdot-}$ radical anions, and ammonium

oxalate (AO) competes effectively for the photoholes h^+ . Dimethyl sulfoxide (DMSO) or suitable metal cations (e.g., Ag^+) also compete for the photoelectrons whenever BaBiO₃ particles are decorated with metal ions. However, the latter was not used in the present study, as they would have altered the nature of the initial photocatalyst and thus its subsequent photocatalytic activity.

In a typical run involving the presence of scavengers, a sample of 50 mg of BaBiO₃ was added to 50 mL of an aqueous phenol solution followed by the addition of 0.20 mmol L⁻¹ of the selected radical scavenger [26]. Experiments were run in four photocatalytic reactors, three of which contained one of the radical scavengers (i.e., IPA or *p*-BQ or AO), while the fourth was used for comparison purposes (control). In all cases, the experimental conditions were otherwise identical as for the determination of the photocatalytic activity of BaBiO₃. For instance, the suspensions were exposed (but covered) to the lamps' heat for no less than 60 min to achieve adsorption–desorption equilibrium and to test for any possible thermal degradation. Thereafter, the suspension was irradiated for 8.5 h, subsequent to which appropriate aliquots were taken from all four reactors, filtered and their phenol content analyzed by liquid chromatography. The measure of the photocatalytic activity (A in % units) in the presence of selected radical scavengers was estimated from Eq. (1).

$$A = \frac{C_o - C_{8.5}}{C_o} \times 100, \quad (1)$$

where C_o and $C_{8.5}$ are the initial ($t=0$) and final (8.5 h) phenol concentrations, respectively.

2.4 Computational details

The electronic structure of BaBiO₃ was determined by a periodic DFT approach using the Local Density Approximation [27] and the Generalized Gradient Approximation with Perdew-Burke-Ernzerhof (PBE) density functionals [28, 29] as implemented in the ABINIT 6.8.3 program [30]. The basis sets employed the form of norm-conserving Troullier-Martins (TM) [31] and relativistic Hartwigsen-Goedecker-Hutter [32] pseudopotentials with the kinetic energy cutoff of 30 and 50 Hartree, respectively. Spin–orbit coupling (SOC) interactions were taken into account through a scalar-relativistic approach. The Brillouin Zone (BZ) was sampled over the automatically generated Γ -point-centered $8 \times 8 \times 8$ Monkhorst–Pack grid of k -points [33]. The electronic band structures were thus computed at 181 k -points along the Γ -A-C-D-D1-E-X-Y1-Z high symmetry path of the monoclinic $C_{2/m}$ BZ. We applied the energy convergence criterion of 10^{-8} Hartree. The resulting band structures and

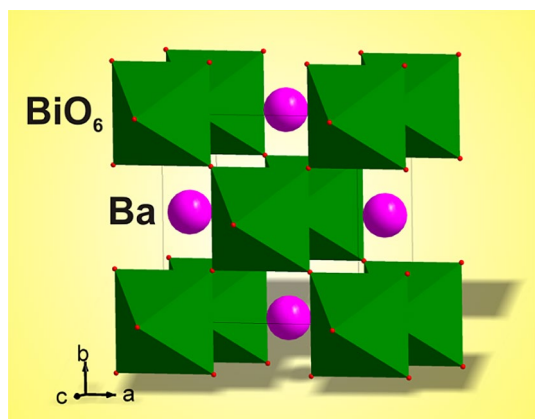


Fig. 2 Cartoon illustrating the $C_{2/m}$ crystal structure of the as-prepared $BaBiO_3$ by the solid-state synthesis

density of states were plotted using the Gnuplot 5.2 software package [34].

3 Results

3.1 Rietveld refinement of the $BaBiO_3$ crystal structure

The earlier refinement by Efremov and coworkers [35] of the crystal structure of $BaBiO_3$ assigned it to the $I_{2/m}$ space group with cell parameters: $a = 6.1736 \text{ \AA}$, $b = 6.1237 \text{ \AA}$, $c = 8.6507 \text{ \AA}$, $\beta = 90.29^\circ$ and $V = 327.038 \text{ \AA}^3$. Initially, we used this assignment to interpret our XRD data of the as-prepared $BaBiO_3$. However, we failed to get a good profile fitting and achieved an R-factor of only 12.38%. Consequently, we indexed our experimental XRD pattern using the Topas 4.2 program [22] that resulted in assigning the structure of $BaBiO_3$ to the $C_{2/m}$ space group with cell parameters: $a = 6.183 \text{ \AA}$, $b = 6.140 \text{ \AA}$, $c = 4.334 \text{ \AA}$, $\beta = 89.82^\circ$, $V = 164.53 \text{ \AA}^3$ that gave a figure of merit of 20.3%. The two cells could not be transformed into each other because of a large difference in cell volumes. The smaller cell with $C_{2/m}$ symmetry was advantageous as the number of refinement parameters was significantly smaller. We utilized the parent cubic P_{m3m} phase of $BaBiO_3$ and distorted it down to the $C_{2/m}$ symmetry using the ISODISTORT internet tool [36] to determine the atomic coordinates. As a result, all ions were located in special positions (see Fig. 2); their coordinates needed no further refinement.

The refinement was stable and gave low R-factors (Table 1, Fig. 3); for example, the R-factor dropped to 9.54% in comparison to that of the $I_{2/m}$ model. Crystal structure tests with the CheckCif internet tool [37] revealed the following missing symmetry elements ‘3’, ‘4’, ‘m’ and suggested the P_{m3m} space group rather than

Table 1 Principal parameters of the processing and refinement of $BaBiO_3$

Compound	$BaBiO_3$
Space group	$C_{2/m}$
$a, \text{ \AA}$	6.1869 (1)
$b, \text{ \AA}$	6.1392 (2)
$c, \text{ \AA}$	4.3349 (1)
$\beta, ^\circ$	89.824 (1)
$V, \text{ \AA}^3$	164.648 (7)
Z	2
2θ -interval, $^\circ$	15–140
$R_{wp}, \%$	9.54
$R_p, \%$	7.39
$R_{exp}, \%$	5.94
χ^2	1.61
$R_B, \%$	3.26

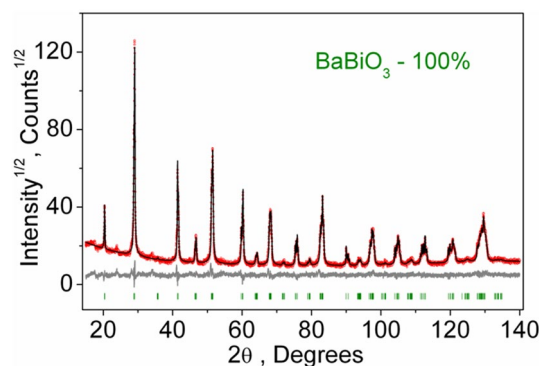


Fig. 3 Difference Rietveld plots of the $C_{2/m}$ (red) versus the P_{m3m} structure (black) of $BaBiO_3$

Table 2 $BaBiO_3$ fractional atomic coordinates and isotropic displacement parameters, \AA^2

	X	Y	Z	B_{iso}
Ba	0	1/2	1/2	0.70 (4)
Bi	0	0	0	0.50 (4)
O_1	1/4	1/4	0	3.0 (2)
O_2	0	0	1/2	3.0 (2)

Table 3 Main bond lengths (\AA) of $BaBiO_3$

Ba— O_1	3.0700 (1)	Ba— O_2^{ii}	3.0935 (1)
Ba— O_1^i	3.0767 (1)	Bi— O_1	2.1790 (1)
Ba— O_2	3.0696 (1)	Bi— O_2	2.1674 (1)

Symmetry codes: (i) $x, y, z+1$; (ii) $-x+1/2, -y+1/2, -z+1$

$C_{2/m}$. However, the main peak splitting in the powder pattern (Fig. 3) showed that a cubic phase was not possible and such peak splitting could only be fitted by a monoclinic unit cell. Therefore, the proposed $C_{2/m}$ unit cell was

considered more appropriate as it fitted well with the XRD pattern.

Tables 2 and 3 report the corresponding atomic coordinates and selected bond lengths, respectively. For further details, see Fachinformationszentrum Karlsruhe [38]; deposition number CSD-2055389.

3.2 Raman spectroscopy

Figure 4 displays both the original Raman spectrum and its baseline-corrected spectrum of the as-synthesized BaBiO₃, which reveals a low-intensity narrow signal with a maximum at 307.4 cm⁻¹ accompanied by a significantly broader signal that spans the 400–600 cm⁻¹ region, which consists of three Gaussian signals at 477.0 cm⁻¹, 510.0 cm⁻¹ and 562.3 cm⁻¹. The spectrum accords with those from Raman spectra of BaBiO₃ reported by others [9, 39].

Assignment of vibrational modes in the Raman spectrum of BaBiO₃ has been somewhat controversial as Ba–O modes in BaO observed by de Waal et al. [40] occur at 513, 477, and 465 cm⁻¹. Nonetheless, the intense signal at 562.3 cm⁻¹ likely corresponds to the A_g breathing mode of BiO₆ octahedra, whereas the signal at 307.4 cm⁻¹ is likely due to the asymmetrical breathing mode of the BiO₆ octahedra [39], although Talha and Lee [41] assigned it to a Bi–O bending mode. A detailed consideration of the broad intense Raman signal at 564 cm⁻¹ reported by Ge et al. [9] and by Talha and Lee [41] also consists of a contribution from a lesser intense signal at 490–492 cm⁻¹ that corresponds to a Bi–O stretching mode.

The Bi–O bond lengths (R ; in Å units) are related to the Raman scattering frequency ν (cm⁻¹) through the empirical Eq. (2) [42]. Thus, the frequencies 307.4 cm⁻¹ and

562.3 cm⁻¹ correspond to Bi–O bond lengths of 2.27 Å and 2.03 Å, respectively, and accord rather well with the Bi–O bond lengths determined earlier using the Rietveld method: 2.1790 Å and 2.1674 Å (see Table 3).

$$\nu = 92760e^{-2.511R}, \quad (2)$$

3.3 SEM and EDX characterization of BaBiO₃

Figure 5 illustrates (a) the SEM images, (b) the EDX spectra, and the elemental mapping (lower panel) of a typical as-synthesized BaBiO₃ particle. SEM reveals no morphological peculiarities (pores, layers and others) for the particle, while the EDX spectrum shows that the ratio Ba/Bi in the particle is very close to being stoichiometric in atom % (Ba_{13.7}Bi_{12.6}O_{73.8} or Ba_{1.1}BiO_{5.9}). The elemental mappings reveal homogeneity among the particles.

3.4 Specific surface area and particle size distribution

The specific surface area of the as-synthesized BaBiO₃ particulates was estimated by the BET method (0.472 m² g⁻¹) and was somewhat less than the 1.2 m² g⁻¹ reported by Tang et al. [2] who prepared BaBiO₃ with a perovskite structure by a soft chemical method, yet an order of magnitude greater than 0.047 m² g⁻¹ reported by Khraisheh et al. [4] who used a conventional solid-state reaction method. Figure 6 illustrates the histogram of the estimated size distribution of BaBiO₃ particles, a large number of which have sizes in the range 409–486 nm.

3.5 Characterization of BaBiO₃ in thin films

The optical properties of BaBiO₃ were examined as thin films deposited on a quartz substrate subsequent to being analyzed first by optical spectroscopy. Figure 7 illustrates the surface of one such film, which shows it to be regular and to cover the whole quartz substrate; photographs of the surface of the quartz substrate and the edge of the film are reported in Fig. S3.

Figure 8 depicts XRD diffractograms of the amorphous quartz substrate and the BaBiO₃ thin film. Evidently, in comparison with the reference XRD pattern of BaBiO₃ (PDF Card No.: 01-074-7522, inset in Fig. 8) the XRD shows the film to be composed of well-formed crystalline BaBiO₃ particles. Consequently, this film proved suitable for investigating the transmittance spectra of BaBiO₃ to assess the bandgap energy.

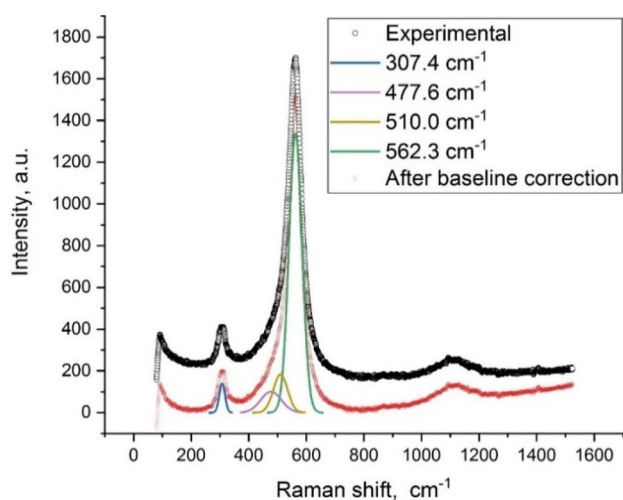


Fig. 4 Raman spectra of the as-synthesized BaBiO₃; the black spectrum refers to experimental data, while the red spectrum refers to experimental data after baseline correction

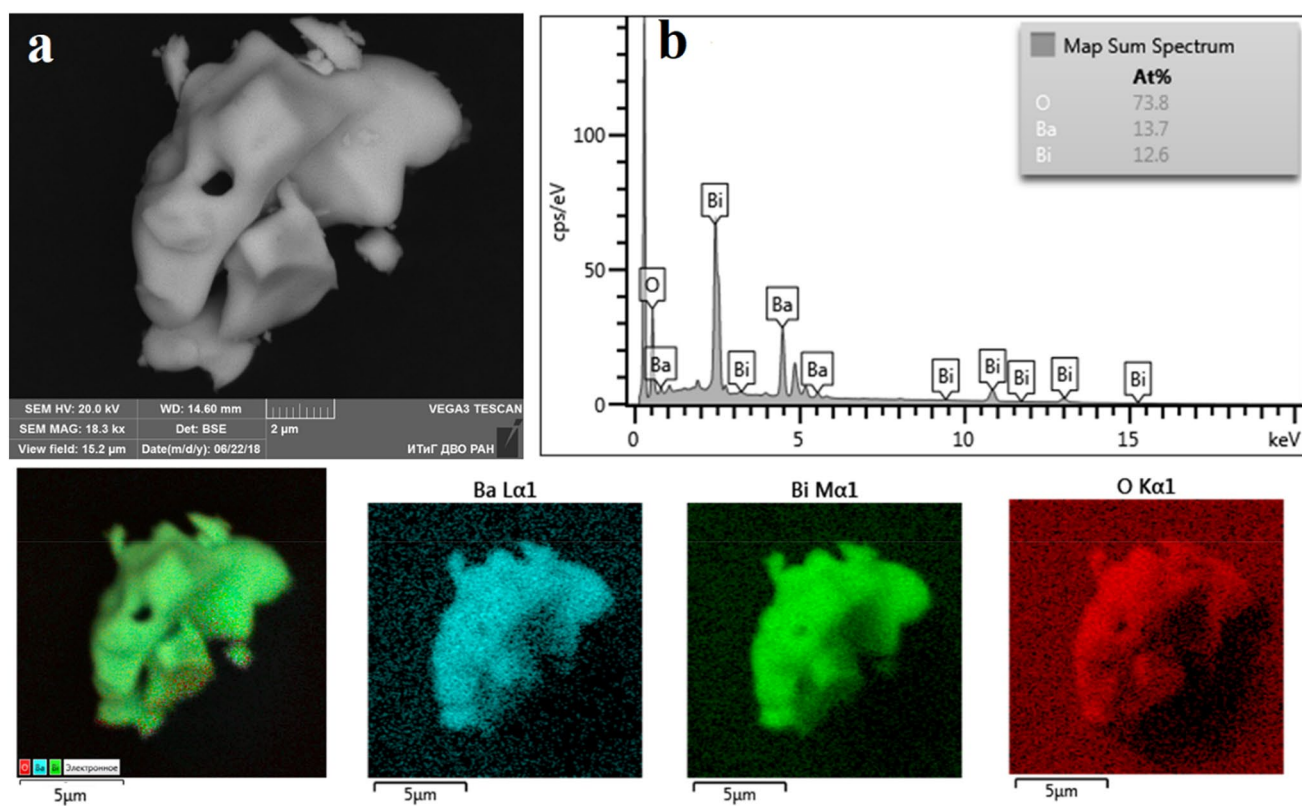


Fig. 5 Typical particle of BaBiO_3 : **a** SEM image; **b** EDX spectrum; lower panel displays the elemental mapping

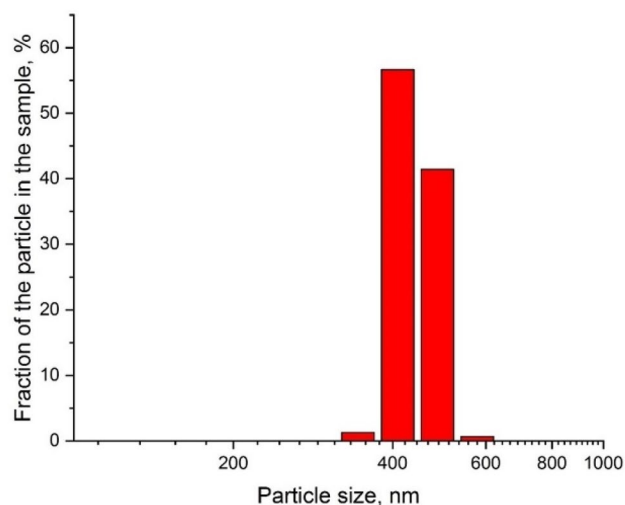


Fig. 6 Histogram displaying the size distribution of the as-synthesized BaBiO_3 particles

3.6 Optical properties of BaBiO_3 in thin films

Ordinarily, an insight into the optical properties of materials implies studies of diffuse reflectance spectra and their subsequent Kubelka–Munk transformation. Unfortunately,



Fig. 7 Optical microscopic image of the surface of a BaBiO_3 thin film

this approach was not appropriate for studies of the BaBiO_3 bismuthate in powdered form. The optical absorption spectrum of powdered BaBiO_3 , derived from the diffuse reflectance spectrum (see Fig. S4), reported as $A = (1 - R)$, showed significant absorption (0.7–0.9 in relative units) in the range from the UV to the Near IR spectral region. Consequently, it precluded the application of the Kubelka–Munk transformation and the subsequent determination of the bandgap energy through the usual Tauc plots. By contrast, the optical

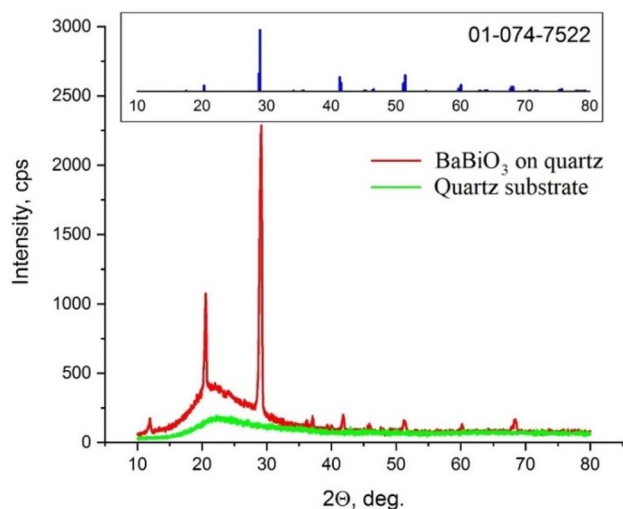


Fig. 8 XRD of the quartz substrate (green) and of a thin film of BaBiO₃ on quartz (red). The inset displays the reference XRD pattern of BaBiO₃

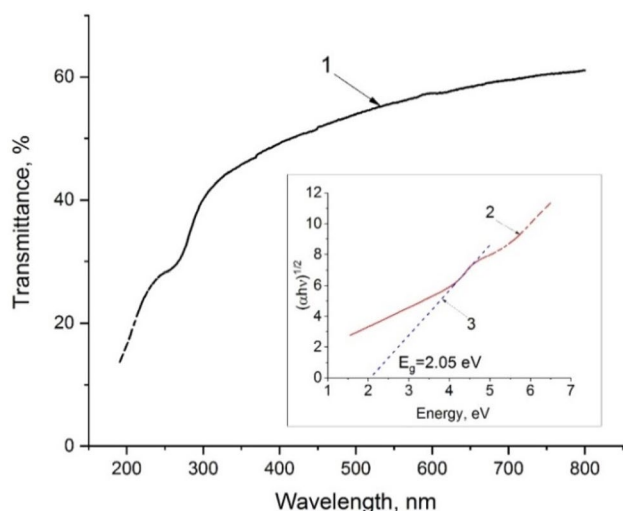


Fig. 9 Optical transmission spectrum of the BaBiO₃ thin film (1). Inset shows the Tauc plot (2) and its linear extrapolation (3) to estimate the bandgap energy

transmission spectrum of the BaBiO₃ thin film, illustrated in Fig. 9 (curve 1), makes it possible to apply the Tauc method to this spectrum (inset Fig. 9), which reveals a bandgap energy E_{bg} for BaBiO₃ of 2.05 eV in accord with earlier reports [1, 2, 11].

3.7 XPS and band structure of BaBiO₃

The XPS spectral bands of the chemical elements in BaBiO₃ presented in Fig. 12 reveal a single Gaussian-like band for Ba (Fig. 10a) at 779.2 eV in line with the bonding energy of

barium in BaO [23], while Bi presents a doublet (Fig. 10b) each of which is also a Gaussian with maxima at 163.9 eV and 158.6 eV corresponding to Bi_{4f5/2} and Bi_{4f7/2}, respectively. All the bismuth species at the BaBiO₃ particle surface are present as Bi³⁺, a point emphasized earlier [1, 11] and conforms with the observation that about half of the bismuths in the bulk of BaBiO₃ particles are present as Bi⁵⁺. In contrast, the oxygen band in Fig. 10c is complex and could be approximated by a sum of two Gaussians with the lower energy signal at 528.6 eV corresponding to O⁻ species, while the higher energy line at 530.6 eV is associated with O²⁻. The presence of O⁻ in alkali earth metal bismuthates noted earlier in our studies [25, 43, 44], as well as by others [45], is thus confirmed.

The energy of the top level of the valence band (E_{VB}) of BaBiO₃ was assessed from the low-energy band edge of the XPS spectrum of O_{2p} displayed in Fig. 11 [46], which yields an $E_{VB} = 1.61$ eV in perfect accord with the value 1.60 eV obtained by Chouhan et al. [11] (see also Fig. S2).

Consequently, the above results and studies by others infer a band structure for BaBiO₃ illustrated in Fig. 12, which emphasizes not only the bandgap of 2.05 eV, but more importantly the position of the flatband potential E_{VB} (= 1.61 eV). An earlier report [47] noted a strong equivalence with the redox potential of 1.59 eV for the half-reaction Bi³⁺ → Bi⁵⁺, in line with the notion that the bulk of BaBiO₃ contains about equal amounts of Bi³⁺ and Bi⁵⁺. Such equivalence will have certain consequences.

It is useful to compare the experimental band structure of BaBiO₃ to the theoretical structure with the latter depending strongly on the approach taken. For instance, the PBE density functional predicts a metallic band structure with zero bandgap, which obviously contradicts experimental observations. Rigorous computations have uncovered the semi-conducting nature of this bismuthate, which predicted the electronic bandgap to vary from 0.84 eV (HSE) to 1.63 eV (GW) [17]. Our own computations suggest that spin-orbit coupling (SOC) interactions are important factors in determining the bandgap of BaBiO₃. Thus, Fig. 13a displays the band structure of the bismuthate's unit cell computed using the LDA + SOC approach. It is evident that the compound features an indirect electronic transition $D1 \rightarrow E$ of 0.72 eV.

As noted above, the bulk of BaBiO₃ contains both Bi³⁺ and Bi⁵⁺ species so that computations with a single unit cell cannot account for such complexity. Consequently, we carried out LDA + SOC computations of the BaBiO₃ band structure with a doubled unit cell illustrated in Fig. 13b as a folded band structure. Comparison with the unfolded band structure reported in Fig. 13c suggests BaBiO₃ to be characterized by both direct and indirect electronic transitions of 1.69 eV ($E \rightarrow E$) and 1.55 eV ($E \rightarrow E/X$), respectively, in good agreement with the data reported recently by Chouhan and coworkers [11]. The greater bandgap energy from the

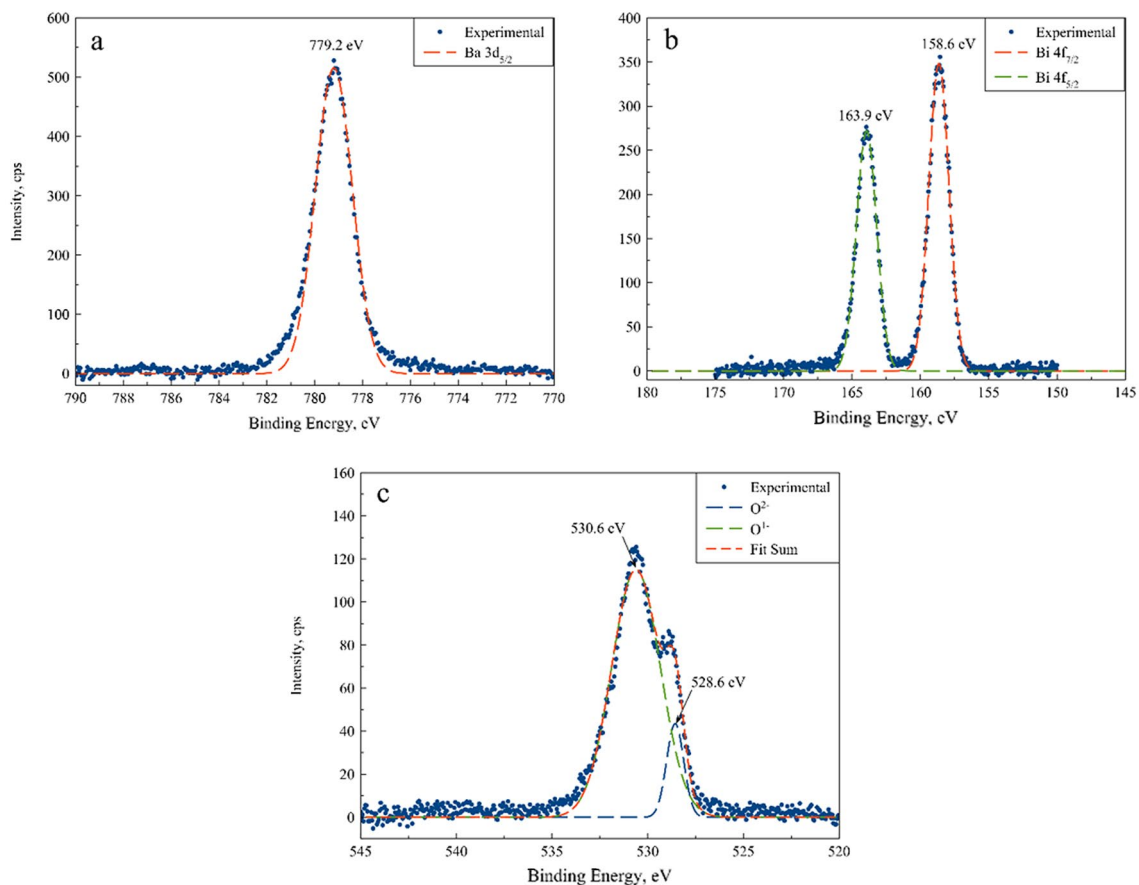


Fig. 10 XPS spectra of the chemical elements constituting BaBiO₃: a Barium; b Bismuth; c Oxygen

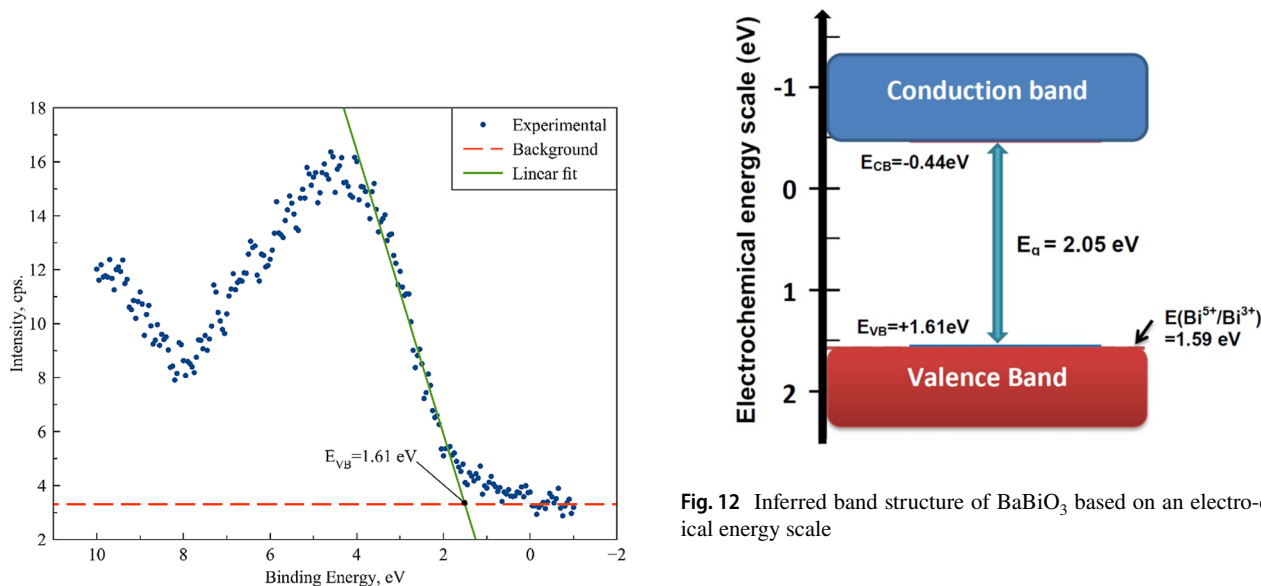
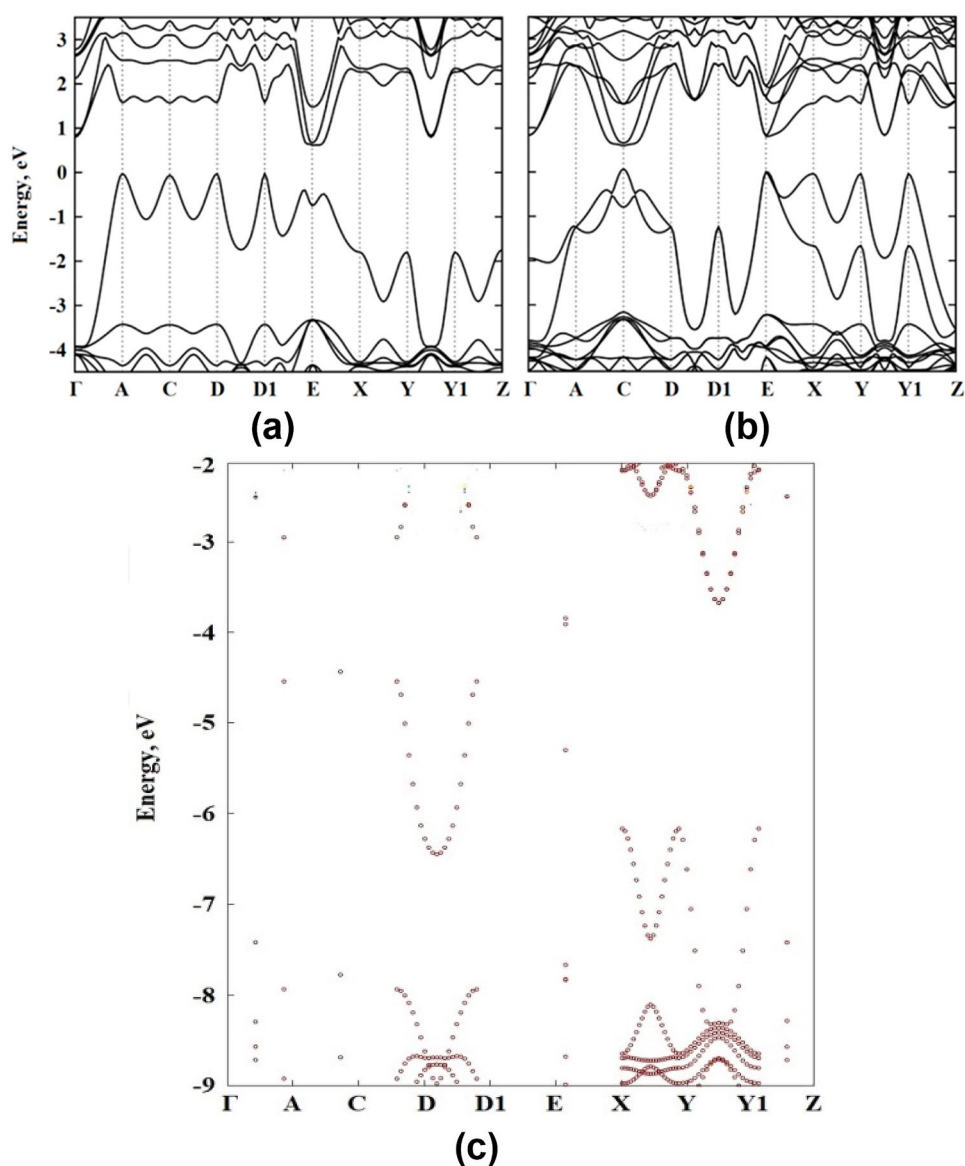


Fig. 11 XPS spectrum of BaBiO₃. The dashed line is a baseline, whereas the solid line is a linear extrapolation of the spectrum's low-energy band edge. The binding energy is given relative to the electrochemical scale in accordance with the methodology proposed in Ref. [44]

Fig. 12 Inferred band structure of BaBiO₃ based on an electro-chemical energy scale

experiment (2.05 eV) might possibly be caused by bandgap widening from thermal lattice expansion under the prevalent experimental conditions. Nonetheless, the bandgaps computed herein correlate well with those reported by Franchini and coworkers [17]. Results obtained from the simulation

Fig. 13 LDA+SOC computed band structure of BaBiO₃: **a** single unit cell, **b** doubled unit cell, **c** unfolded band structure of the doubled unit cell



also explain why an indirect optical transition appeared in our study of the optical properties of thin films. To the extent that its energy is somewhat less than the energy of the direct transition explains why we failed to detect the direct optical transition experimentally.

3.8 Photocatalytic activity of BaBiO₃

Figure 14 reports the time course of the photocatalytic degradation of phenol in aqueous media in the presence of BaBiO₃, while the inset shows the zero-order degradation rates for the control experiment in the absence of BaBiO₃, and for comparison also reported is the degradation rate of the reaction catalyzed by Evonik's P25 TiO₂. Evidently, under otherwise comparable experimental conditions, the reaction proceeds somewhat faster in the presence of the visible-light-active

BaBiO₃ photocatalyst than in the presence of the otherwise UV-light-active TiO₂ photocatalyst.

Experiments conducted in the presence of selective radical scavengers disclosed some of the mechanistic details of the degradation of phenol in the presence of BaBiO₃. The relative degradation efficiencies are displayed in Fig. 15, which shows that the usage of p-benzoquinone and ammonium oxalate lowered the rate of phenol degradation more than threefold. The active species that played a key role in this photocatalytic reaction appear to be the superoxide radical anions O₂^{•-} and the photoholes h⁺.

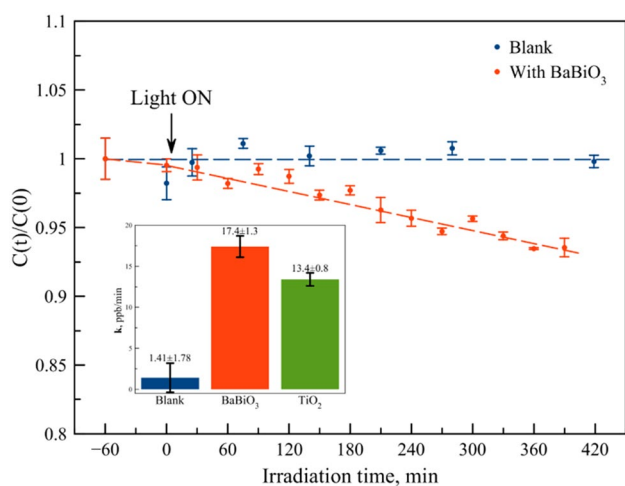


Fig. 14 Kinetics of the photocatalytic degradation of phenol with and without the presence of the BaBiO_3 photocatalyst. Inset reports the resulting zero-order degradation rates. For comparison, the degradation rate of phenol in the presence of Evonik's P25 TiO_2 is also reported

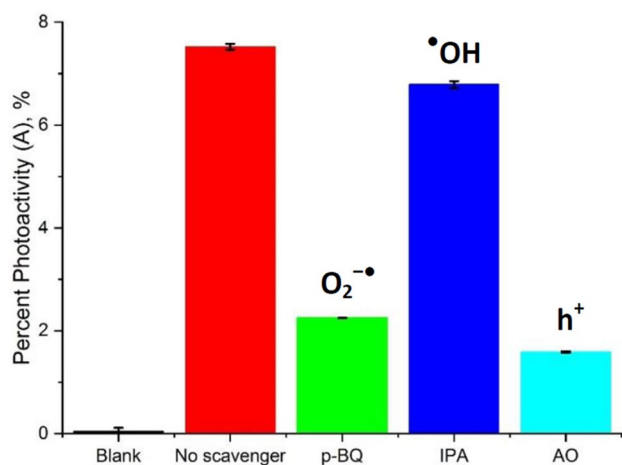


Fig. 15 Percent photocatalytic activity of BaBiO_3 in the photocatalyzed degradation of phenol in the presence of various selective radical scavengers

4 Discussion and concluding remarks

We begin by addressing some of the inconsistencies arising when the experimental data on the photocatalytic activity of BaBiO_3 are compared to its band structure (Fig. 1). For instance, according to Tang and coworkers [2], acetaldehyde can be degraded photocatalytically in the presence of BaBiO_3 , yet the redox potential of this reaction is -0.197 eV [48]. However, given the E_{CB} of BaBiO_3 of -0.18 V reported by these authors, the reaction is energetically unfavorable. Such a contradiction is precluded

if the band structure of BaBiO_3 were the one inferred in the present study, which places the position of E_{CB} at -0.44 eV, enough to allow the photocatalytic degradation of acetaldehyde to occur. Likewise, Kraishieh et al. [4] carried out the photocatalytic conversion of CO_2 to CH_4 in the presence of BaBiO_3 , a process that requires a redox potential of -0.24 V [49]. However, to the extent that the E_{CB} potentials reported by Namatame et al. [1] and Tang et al. [2] are -0.20 V and -0.18 V, respectively, such a conversion should not take place, unless of course the E_{CB} were -0.44 eV as in the present study. For their part, Ge et al. [9] proposed the use of BaBiO_3 for the photocatalytic water splitting process. However, the band structure of BaBiO_3 reported in their study (see Fig. 1) precluded this process from occurring as the E_{VB} of 0.625 V reported by these authors is insufficient for overcoming the potential barrier of the oxidation half-reaction of oxygen formation that requires 1.23 V. The band structure of BaBiO_3 suggested in the present study avoids these conundrums, as the positions of both E_{VB} (1.61 V) and E_{CB} (-0.44 V) are sufficient for the oxidation and reduction half-reactions to occur that would lead to the formation of oxygen and hydrogen, respectively.

Accordingly, BaBiO_3 was characterized and its band structure determined (Fig. 12) that bears close resemblance to the structure reported earlier by Chouhan and coworkers [11]. This bismuthate demonstrated a relatively higher photocatalytic activity than the well-known Evonik P25 TiO_2 toward the degradation of phenol in aqueous media. The usage of selective radical scavengers helped to clarify some details of the photocatalytic accomplishment of BaBiO_3 that were not addressed in previous studies. The use of radical scavengers has inferred a plausible mechanism in the photocatalytic degradation of phenol in aqueous media.

Figure 16 summarizes our current understanding of the photocatalytic mechanism of BaBiO_3 . Upon irradiation with

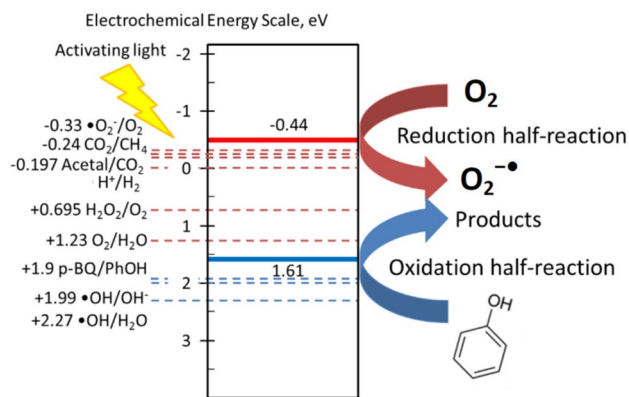


Fig. 16 Energetic scheme of the electrochemical reactions potentially photocatalyzed by BaBiO_3

UV/Visible light, the pair of charge carriers—photoelectrons e^- and photoholes h^+ —are generated with the photoelectrons being trapped by the oxygen dissolved in the aqueous solution and adsorbed on the BaBiO₃ surface to yield superoxide radical anions O₂^{•-}. The subsequent attack of phenol by this radical concludes the reduction half-reaction.

The photocatalytic tests conducted in the presence of selective radical trapping agents also infer that the photoholes h^+ played a non-insignificant role, even though the direct attack of the phenol by photoholes is precluded toward an oxidative half-reaction as the redox potential for such a process is 1.90 V [50], greater than the flatband potential E_{VB} . For the same reason, the hydroxyl radical •OH cannot participate directly in the photocatalytic process as the redox potential of its formation is greater than that of E_{VB} . Accordingly, secondary electrochemical reactions involving photoholes h^+ may be implicated. For instance, the formation of oxygen from water necessitating a redox potential of 1.23 eV, or formation of hydrogen peroxide [2] from the dissolved oxygen in the aqueous medium that requires 0.695 eV are potentially possible, in which case the hydrogen peroxide formed could directly participate in the oxidation of phenol.

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Declarations

Conflict of interest The authors declare that they have no conflict of interest.

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