



Magnetic anisotropy and core-shell structure origin of the biogenic ferrihydrite nanoparticles



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ABSTRACT

Ferrihydrite is a low-crystalline nanoscale matter. The uncompensated magnetic moment of the ferrihydrite caused by the antiferromagnetic ordering of the magnetic moments of iron atoms and leads to the magnetic properties very similar to those of ferro- and ferrimagnetic nanoparticles. In this study, we investigated the biogenic ferrihydrite nanoparticles with the narrow size distribution and an average diameter of ≈ 2 nm obtained by the bacteria life cycle. The features caused by the surface effects and the inhomogeneous structure of ferrihydrite have been examined in the temperature range of 4–300 K using Mössbauer spectroscopy and magnetometry. Based on the Mössbauer data, we identified the superparamagnetic blocking temperature at the temperature of 30 K for the largest ferrihydrite particles. We established that the exceptional magnetic anisotropy of ferrihydrite ($K_V = 1.2 \cdot 10^5$ erg/cm³ and $K_S = 0.1$ erg/cm²) is reached because of the highly developed ferrihydrite nanoparticles' surface. According to the Mössbauer data, we propose a core-shell structural model of the biogenic ferrihydrite particles. We found that the size of the dense core depends on the particle size. The well-crystallized core is formed only for nanoparticles larger than ≈ 2 nm, whereas smaller particles consist entirely of a matter with a lower density of iron atoms.

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1. Introduction

Ferrihydrite plays a decisive role in the circulation of iron and controls the iron concentration in aqueous systems. This mineral with the nominal formula $\text{Fe}_2\text{O}_3 \times n \cdot \text{H}_2\text{O}$ precipitates as a product of nucleation of hydrolyzed ferric ions with increasing pH [1]. The discovery of ferrihydrite on Mars [2,3] indirectly confirmed the presence of water on this planet. A huge surface makes ferrihydrite a good sorbent of heavy metals [4,5]. This mineral is of crucial importance for microorganisms' life [5,6]. Ferrihydrite can appear also in the bacterial life cycle [7]. Then, the size of crystallized biogenic ferrihydrite particles is no more than ≈ 10 nm [8–17,17–19]. This mineral is involved in the vital activity of higher

animals, including humans. In living organisms, ferrihydrite forms in the core of a ferritin complex consisting of a protein shell capsule with the ordered hydrated iron oxide core. In the most well-studied horse spleen ferritin, the ordered ferrihydrite core is located inside the protein shell with outer and inner diameters of 12 and 5–8 nm, respectively [8,10]. It is noteworthy that the organic shell makes the magnetic interparticle interactions negligible [11].

The conditions for the formation of ferrihydrite significantly affect the physical properties of its nanoparticles, concerning particularly its crystallinity and interactions between individual particles [12–14]. The crystal structure of ferrihydrite and its chemically synthesized aggregates was studied in Refs. [6,15,16]. The obtained X-ray data made it possible not only to refine the crystal structure and chemical formula of ferrihydrite, but also to determine the degree of its atomic disorder. Study of aggregation of ferrihydrite nanocrystals showed that the formation of aggregates cannot be described by conventional growth models [14]. A nano-sized ferrihydrite particle can be presented by a model with different surface (shell) and internal (core) positions of iron atoms

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[9,18]. As was found in Ref. [19], the structure of ferrihydrite polyhedra near the surface, which is depleted in iron and rich in OH-groups, strongly differs from the structure of the particle core [9,19]. As it was mentioned in Ref. [19], the ferrihydrite particle surface is depleted in iron and rich in OH-groups, that strongly differs from the structure of the particle core. Due to such a feature of the ferrihydrite surface formation, the surface enthalpy is much lower in comparison with other iron oxides and hydroxides [9,10]. However, the particle itself remains chemically stable. The established effect of depletion of the nanoparticle surface [19] manifests itself in the low nanoparticle fractal dimension (≈ 2.82) determined by the small-angle X-ray scattering technique [20,21].

The magnetic moments of iron atoms in ferrihydrite are antiferromagnetically ordered; according to the neutron diffraction data, the Néel temperature of the material is ≈ 350 K [22]. In turn, the imperfection of the ferrihydrite crystal structure leads to the occurrence of an uncompensated magnetic moment in nanoparticles [23–29]. This fact subsumes ferrihydrite to the class of antiferromagnetic (AFM) nanoparticles [8,11,30]. The uncompensated magnetic moment of AFM nanoparticles determines their magnetic properties, which are very similar to those of single-domain ferromagnetic (FM) and ferrimagnetic nanoparticles [11,22,24–30]. In AFM nanoparticles, the magnetic properties exhibit the features typical of those of ordinary FM nanoparticles, including the transition from the unblocked or superparamagnetic (SP, in case of the uncompensated magnetic moments of particles) state to the blocked state with a decrease in temperature (or increase in size at $T = \text{const}$) and the dependence of the SP blocking temperature on the ratio between the characteristic particle relaxation time and the detection time determined by a measurement procedure used. The uncompensated magnetic moment of AFM nanoparticles can attain several hundreds of Bohr magnetons [23–38], which is comparable with the magnetic moment of FM oxide particles several nanometers in size [39–41]. The latter opens wide prospects for use of AFM nanoparticles, owing to their magnetization impacted by their nanometer dimensions. Possible medical applications of ferrihydrite based on its magnetic properties were described in Refs. [42,43].

The combination of the above-mentioned factors (AFM ordering, uncompensated magnetic moment, developed surface, surface depletion effect [9,18,19], and possible magnetic interparticle interactions [8,44–46]) determine fairly complex behavior of the magnetic properties of ferrihydrite. Nevertheless, several bright effects directly reflecting the core-shell structure of ferrihydrite nanoparticles were established using the quasi-static magnetization [25,27] and ferromagnetic resonance [47] measurements.

In this work, we applied Mössbauer spectroscopy and the quasi-static magnetization measurements to study the features of the manifestation surface and “core” iron atoms magnetism in ferrihydrite nanoparticles. The presence of a polysaccharide shell surrounding nanoparticles of investigated biogenic ferrihydrite [48] minimizes magnetic interactions between particles. This allows us to ignore the magnetic interactions between particles from the analysis, as in the case of horse spleen ferritin. As a result, we separated the two contributions in the magnetic behavior of biogenic ferrihydrite: 1) the core-shell structure contribution, and 2) the effect of the transition to the SP state.

2. Experimental

The *Klebsiella oxytoca* strain used was extracted from the sapropele of the Borovoe Lake, Krasnoyarsk Territory, Russia. Microorganisms were seeded on an agarized medium and grown under anaerobic conditions [49,50]. After the repeated ultrasonic treatment of precipitated bacteria, centrifugation, and washing, a stable

nanoparticle sol in an aqueous solution was prepared and then dried. The obtained nanoparticle powder was the object of study.

The electron microscopy and microdiffraction investigations were carried out on a Hitachi HT7700 transmission electron microscope at an accelerating voltage of 100 kV. Specimens were prepared by shaking the nanoparticle powder in alcohol in an ultrasonic bath and depositing the obtained suspension onto support meshes with a perforated carbon coating.

The temperature dependences of the magnetization $M(T)$ were measured on a SQUID magnetometer [51] in the zero external field cooling (ZFC) and field cooling (FC) modes.

The Mössbauer spectra of the investigated sample were obtained on an MS-1104Em spectrometer (Research Institute of Physics, Southern Federal University) in the transmission geometry with a $\text{Co}^{57}(\text{Rh})$ radioactive source in the temperature range of 4–300 K using a CFSG-311-MESS cryostat with a sample in the exchange gas based on a closed-cycle Gifford-McMahon cryocooler (Cryotrade Engineering). The spectra were processed by varying the entire set of hyperfine parameters by the least-squares' method in the linear approximation. The spectra were simulated by Lorentz lines with regard to the broadenings caused by the magnetic and crystalline inhomogeneity of the nanoparticles within the sample.

3. Results and discussions

3.1. Microstructure

Fig. 1 shows typical transmission electron microscopy (TEM) data for the obtained nanoparticles and their size distribution $f(d)$. The average particle diameter $\langle d \rangle$ was found to be 2 nm. Inset in Fig. 1 presents a microdiffraction pattern characteristic of ferrihydrite nanoparticles [52–54]. One can see two diffuse reflections with interplanar spacings of 1.5 Å and 2.6 Å.

3.2. SP blocking for the magnetic measurements and Mössbauer spectroscopy

Fig. 2a shows the $M(T)$ dependences of the investigated ferrihydrite nanoparticle sample that were recorded in the ZFC mode and in the FC mode in fields of 1 Oe and 1 kOe. The $M(T)_{\text{ZFC}}$ dependences contain pronounced maxima, which shift toward lower temperatures ($T_{\text{max}} \approx 13.7$ K at $H = 1$ Oe and $T_{\text{max}} \approx 9.5$ K at $H = 1$ kOe) with increasing field. Near T_{max} , the $M(T)_{\text{ZFC}}$ and $M(T)_{\text{FC}}$ dependences start diverging; the divergence increases with decreasing temperature. This behavior is typical of the SP blocking processes.

Fig. 2b presents the $M(T)$ dependences recorded in the ZFC mode and in the FC mode in a field of 100 Oe. It is well-known that the derivative $\frac{d(M(T)_{\text{FC}} - M(T)_{\text{ZFC}})}{dT}$ reflects the blocking temperature distribution function $f(T_B)$, the maximum of which can be taken as an average value $\langle T_B \rangle$ [55–57]. The $d(M(T)_{\text{FC}} - M(T)_{\text{ZFC}})$ dependence and its derivative $f(T_B)$ are shown in Fig. 2b. Below, we use these data¹ to make a comparison with the $f(d)$ dependence based on the TEM results.

The Mössbauer spectra of the investigated biogenic ferrihydrite sample are shown in Fig. 3. At room temperature, the spectrum is a quadrupole doublet, which is explained by the SP behavior of ferrihydrite nanoparticles. The room-temperature spectrum reveals three nonequivalent positions of ferric iron in the octahedral

¹ For an external field $H = 100$ Oe, the value of T_{max} (≈ 12.6 K) does not differ much from T_{max} for an “extremely small” field (1 Oe), and at the same time, the signal-to-noise ratio provides an acceptable error in determining the derivative for an external field $H = 100$ Oe.

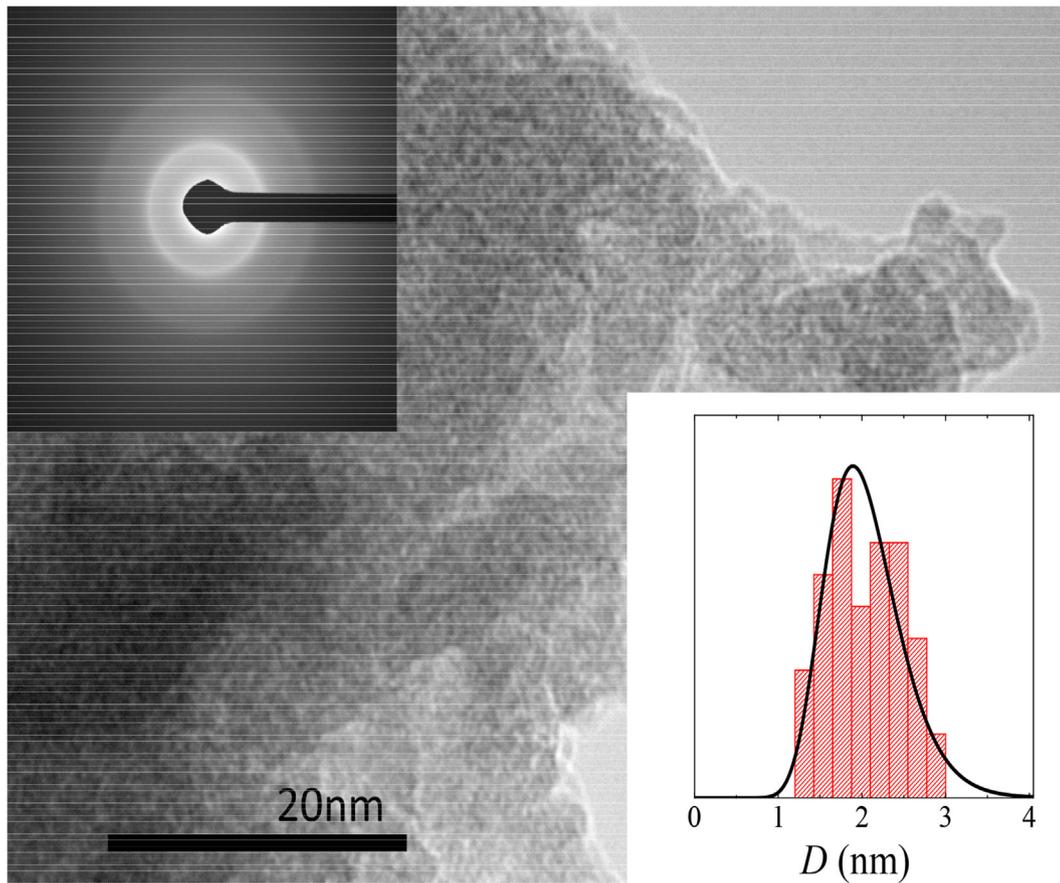


Fig. 1. TEM image of biogenic ferrihydrite nanoparticles. Inset: particle size distribution and microdiffraction pattern.

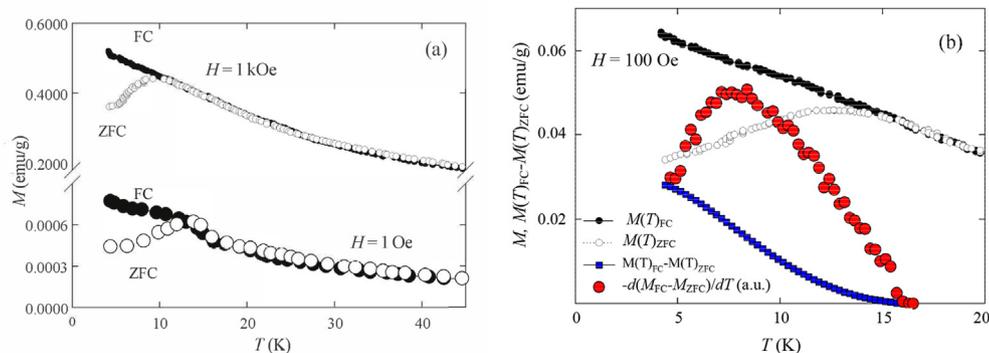


Fig. 2. Temperature dependences of magnetization measured under ZFC and FC conditions in fields of 1 Oe, 1 kOe (a), and 100 Oe (b). In (b) also plotted the difference $M(T)_{FC} - M(T)_{ZFC}$ and the derivative $\frac{d(M(T)_{FC} - M(T)_{ZFC})}{dT}$.

environment D1, D2, D3 (see Table 1), which are characteristic of ferrihydrite [1,54,57–60]. Their parameters are essentially temperature independent and given only for temperature of 300 K and 40 K in Table 1. A slight change of the isomer shift (IS) value is due to the second-order Doppler shift. Below the temperature of 30 K, a six-line Zeeman components are well resolved (labeled by S1 and S2 in the Table 1). This temperature is much lower than the temperature of the AFM ordering of ferrihydrite [22]. Therefore, the occurrence of the magnetically split component in the Mössbauer spectrum can be considered in terms of blocking of the magnetic moments of particles (the uncompensated moments for AFM nanoparticles) [61]. Obviously, 30 K is the blocking temperature of

the largest particles. With decreasing temperature, the SP fractions and the magnetically split parts of the spectrum are redistributed, which can be reasonably attributed to the blocking of smaller particles.

We note that, in some cases, the transition from the SP to blocked state is accompanied by the occurrence of a relaxation component in the Mössbauer spectra of nanoparticle systems, which can be approximated by a single broad line (labeled by L1 in Table 1 at 10 K) [12,61]. In our case, its contribution is negligible (see Table 1 and Fig. 3). This can be explained by the absence of interparticle magnetic interactions, which are prevented by the polysaccharide shell of biogenic ferrihydrite nanoparticles

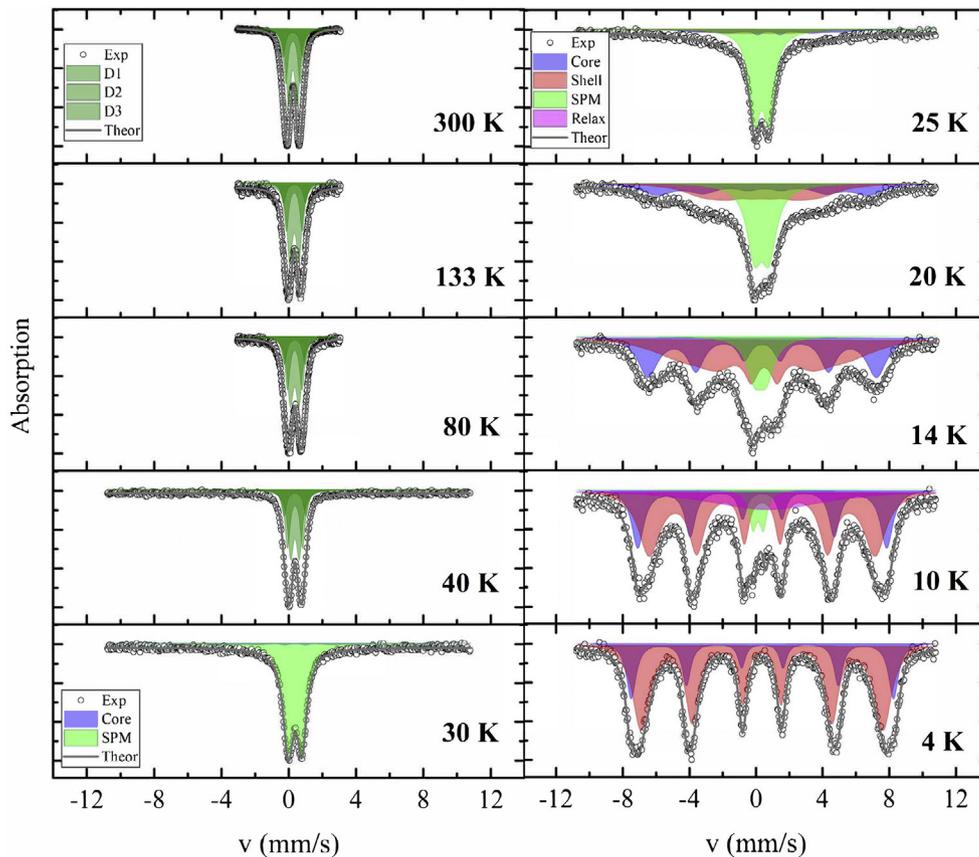


Fig. 3. Mössbauer spectra of the biogenic ferrihydrite sample at different temperatures within the range of 4–300 K. The partial components obtained by processing of the spectra are shown (see the text and Table 1).

[47,60,62]. This makes it possible to reliably distinguish the SP spectral component (doublet D) in the temperature range of 4–30 K against the background of particles with a blocked magnetic moment, which are reflected by a sextet with broadened lines. The temperature dependences of the intensities of the magnetically split (sextets S1 and S2) and SP (doublet D) spectral components are shown in Fig. 4. Due to the particle size distribution, the hyperfine structure of the Mössbauer spectra smoothly collapses. A fairly narrow (4–30 K) temperature range (as compared, for example, with the results of [61,63,64]) of the transition from the blocked to SP state follows from the narrow size distribution of the investigated ferrihydrite nanoparticles (Fig. 1).

Let us consider the process of nanoparticle blocking in more detail using the Mössbauer spectrometry and magnetic measurement data. The magnetic moment of nanoparticles of a certain size is blocked at a specific temperature, which can be determined by the Néel–Brown relationship (1):

$$T_B = \frac{K_{\text{eff}}V}{\ln(\tau/\tau_0)k_B} \quad (1)$$

In this equation, K_{eff} is the effective magnetic anisotropy constant, V is the particle volume, τ_0 is the characteristic particle relaxation time lying conventionally between 10^{-13} – 10^{-9} s [11], and τ is the characteristic measuring time, which is $\approx 10^2$ s for the quasi-static magnetic measurements, while in Mössbauer spectroscopy, is determined by the characteristic lifetime of ^{57}Fe nuclei in the excited state and amounts $\tau_0 \approx 2.5 \cdot 10^{-8}$ s [65]. The effective magnetic anisotropy is frequently expressed as [66–68].

$$K_{\text{eff}} = K_V + \frac{6K_S}{d} \quad (2)$$

Eq. (2) takes into account the contributions of the bulk (K_V) and surface magnetic anisotropy. The latter is induced by the large fraction of spins with the incomplete environment [69,70] and can be characterized by the surface anisotropy constant K_S .

The magnetic measurements illustrated in Fig. 2b yield the blocking temperature distribution function

$$f(T_B) \approx -\frac{d(M(T)_{\text{FC}} - M(T)_{\text{ZFC}})}{dT} \quad (3)$$

Considering the normal particle size distribution in ferrihydrite sample, the temperature dependence of the SP component fraction (Fig. 4) can be described by the variable probability integral [61,71].

$$\Phi(x) = \frac{2}{\sqrt{2\pi}} \int_0^x e^{-\frac{t^2}{2}} dt \quad (4)$$

here, $\Phi(x)$ is the blocking temperature distribution function: $\Phi(x) \propto f(T_B)$ (Fig. 4). And integration carried by all existing blocking temperature due to particle size distribution. t means the normalized deviation of the blocking temperature from the $\langle T_B \rangle$. Then, differentiation of function (4) gives us a distribution $f(T_B)$ in the sample. Obviously, the average values $\langle T_B \rangle$ and the distribution functions $f(T_B)$ (Figs. 2b and 4) obtained by different experimental techniques are different because of the difference in the characteristic times τ , according to Eq. (1). However, at the correct K_{eff} and

Table 1

Mössbauer spectra parameters. IS is the isomer shift relative to α -Fe, QS is the quadrupole splitting, H_{hf} is the hyperfine field, W is the full width of the line at half-maximum, dW is the degree of nonuniformity of the linewidth, and A is the relative fraction in the spectrum.

	IS , mm/s ± 0.005	H_{hf} , kOe ± 2	QS , mm/s ± 0.01	W , mm/s ± 0.01	dW , mm/s ± 0.01	A , a.u. ± 0.03	Description
4 K							
S1	0.490	490	0.00	0.50	0.21	0.26	Core
S2	0.468	452	0.00	0.49	0.75	0.74	Shell
10 K							
S1	0.480	465	0.00	0.56	0.34	0.25	Core
S2	0.475	422	0.00	0.53	1.06	0.49	Shell
L1	0.680	0	0.01	8.88	–	0.21	Shell
D	0.435	–	0.62	0.56	–	0.05	SP
14 K							
S1	0.475	426	0.00	0.79	0.34	0.35	Core
S2	0.507	303	0.00	0.29	1.06	0.52	Shell
D	0.367	–	0.72	1.24	–	0.13	SP
20 K							
S1	0.553	395	–0.24	0.79	0.74	0.28	Core
S2	0.419	216	0.00	0.29	1.40	0.33	Shell
D	0.445	–	0.88	1.24	–	0.38	SP
25 K							
S1	0.480	264	0.00	0.79	2.94	0.16	Core
S2	0.478	151	0.00	3.05	0.00	0.15	Shell
D	0.462	–	0.83	0.83	–	0.69	SP
30 K							
L1	0.423	0	0.00	0.42	2.41	0.06	Core
D	0.476	–	0.83	0.67	–	0.94	SP
40 K							
D1	0.476	–	0.53	0.43	–	0.46	SP1
D2	0.480	–	0.90	0.34	–	0.37	SP2
D3	0.478	–	1.28	0.36	–	0.17	SP3
300 K							
D1	0.360	–	0.53	0.34	–	0.47	SP1
D2	0.358	–	0.86	0.32	–	0.35	SP2
D3	0.354	–	1.23	0.34	–	0.17	SP3

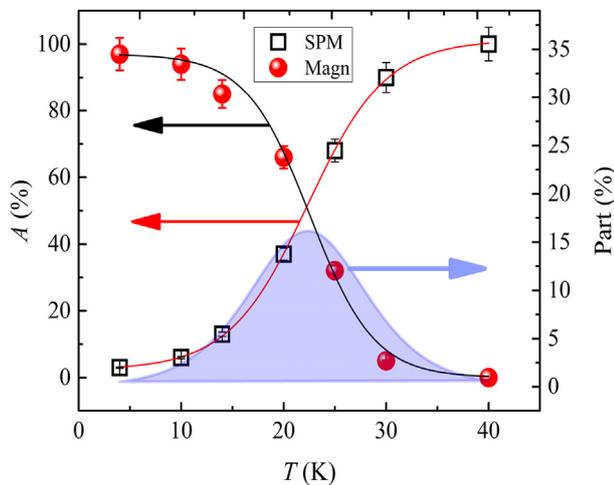


Fig. 4. Temperature dependences of the populations of the doublet (SP) and magnetically split (sextet – Magn) components of the Mössbauer spectrum. The fill area shows the blocking temperature distribution of nanoparticles.

τ_0 values, the recalculation of the $f(T_B)$ dependences using Eq.(1) in the $f(d)$ distribution function should yield the coincidence of the data obtained by different techniques, as well as the coincidence with the distribution based on the TEM data (Fig. 1).

Using Eqs. (1) and (2), we obtain

$$K_V d^3 + 6K_S d^2 = \ln(\tau / \tau_0) k_B T_B \quad (5)$$

Due to the relation (1) there is a direct dependence between the

blocking temperature and the corresponding value, i.e. particle diameter. Therefore, solving numerically cubic equation (5) for each blocking temperature in accordance with the early obtained $f(T_B)$ (see Figs. 2b and 4), we can obtain the particle size distribution $f(d)$ from the $f(T_B)$ dependence.

The best agreement between the calculated $f(d)$ dependences obtained from the magnetic measurements and Mössbauer spectroscopy data and the TEM results was obtained at $K_V = 1.2 \cdot 10^5$ erg/cm³ and $K_S = 0.1$ erg/cm², and $\tau_0 = 10^5$ s. Fig. 5 shows the calculated $f(d)$ dependences along with the TEM data. It can be seen that these dependences agree well with each other. It is noteworthy that the use of Eq. (1) and the effective anisotropy constant for average-size particles ($K_{eff}(d = 2 \text{ nm}) \approx 3.1 \cdot 10^5$ erg/cm³) noticeably narrow the bulb of the calculated $f(d)$ dependences. The results obtained confirm the need for taking into account the surface anisotropy contribution in ferrihydrite, similar to other oxide nanoparticles [37,72–74]. The K_V and K_S values are consistent with the results obtained for ferrihydrite in Refs. [44,46,58].

3.3. Manifestation of the core-shell structure in the Mössbauer spectra

Thus, according to the Mössbauer spectroscopy data, the magnetic moments of particles are blocked in the temperature range from 30 to 4 K. In the SP state (> 30 K), the doublets correspond to three nonequivalent positions of ferric iron in the octahedral environment characteristic of ferrihydrite (Table 1). However, during the transition to the blocked state, the magnetically split part of the spectrum (sextet) should reflect the features characteristic of the inhomogeneous structure of the nanoparticles

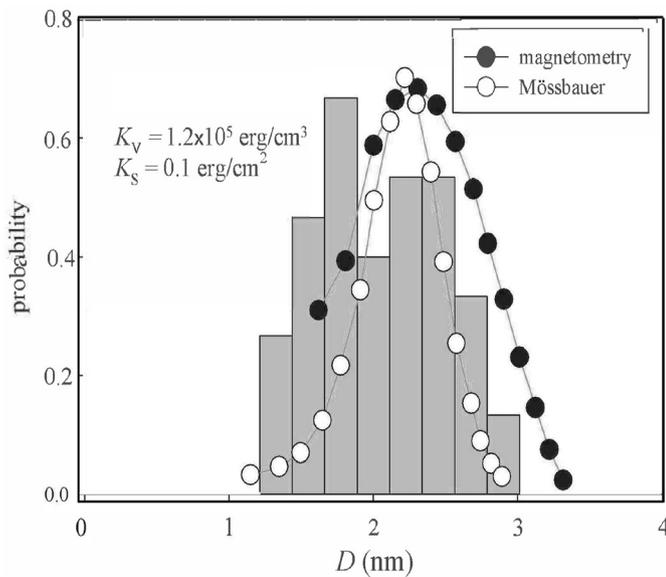


Fig. 5. The hyperfine field at four nonequivalent iron positions as a function of the external magnetic field for various temperatures in the 4–280 K temperature range. At 4 K, the open symbols show the interpolation of the experimental data by a linear function.

themselves [10]. In this approximation, the Mössbauer spectrum for the blocked state can consist of at least two subspectra S1 and S2, which should represent the averaged parameters of nonequivalent magnetic states of iron [10,60]: S1 in the core and S2 in the shell. This approach was considered during fitting procedure of the magnetically split spectral component in the temperature range of 4–30 K. The hyperfine parameters of the spectra based on this fitting model are given in Table 1. The temperature evolution of the populations of the corresponding subspectra is shown in the inset in Fig. 6. It is noteworthy that the population of spectrum component S1 responsible for iron atoms in the particle core remains

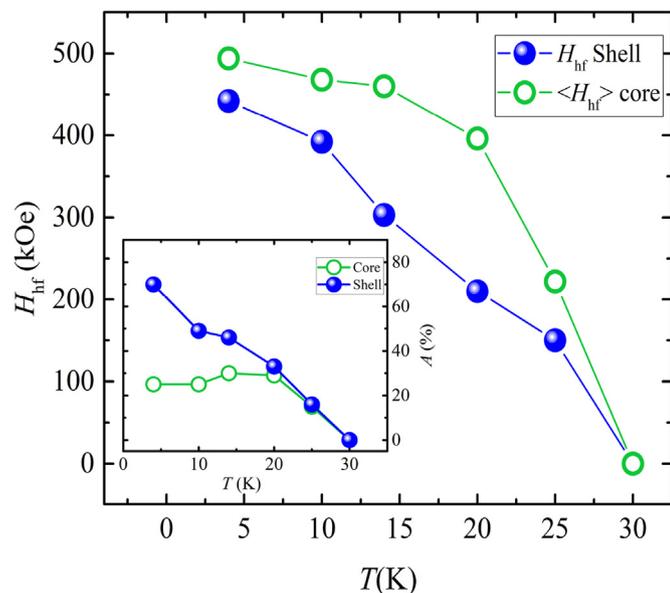


Fig. 6. Temperature dependences of populations A (the ordinate axis is the left-hand scale) and hyperfine field H_{hf} (the ordinate axis is the right-hand scale) of internal states S1 (core) and surface states S2 (shell) of ^{57}Fe in the investigated ferrihydrite obtained by processing of the Mössbauer spectra.

almost constant in the range of 4–25 K. On the other hand, the sextet S2 area, which is taken by us for the surface states of iron atoms, changes quite strongly. This can be related to the fact that, with increasing temperature, we have a complex transition of particles to the SP state, which, in addition to the effect of the particle size distribution (Fig. 5), reflects the features of the inhomogeneous structure of ferrihydrite particles.

The differences between the formation of the core and surface magnetic structures can be clearly seen in the temperature dependence of the hyperfine fields $H_{hf}(T)$ (Fig. 6). On the particle surface, the bond breaking occurs, which usually leads to the reduction of the iron coordination and, in the hydrated state, to the formation of new bonds upon filling of coordination spheres on the surface. Hence, the hyperfine parameters (hyperfine field) of ^{57}Fe nuclei for sextet S2 appear somewhat underestimated due to the uneven arrangement of magnetic neighbors as compared with those in the particle core (S1). The magnetic correlation radius decreases with increasing temperature; one can see that a decrease in the hyperfine field is more pronounced for the iron surface states, which can be reasonably attributed to the weakening of magnetic interactions with increasing distance from the particle center.

Let us consider the temperature evolution of the ratio A_{S2}/A_{S1} between the populations of subspectra S2 and S1 (see the inset in Fig. 7). The A_{S2}/A_{S1} dependence is a linear function of temperature. This, in fact, is the ratio between volumes V_{shell} of the surface states and volumes V_{core} of the core for all particles blocked at a specific temperature: $A_{S2}/A_{S1} \approx V_{shell}/V_{core}$ at $T = \text{const}$. From general considerations, we can write

$$\frac{V_{core}}{V_{shell}} \approx \frac{R^3}{(R + \Delta r)^3 - R^3} \quad (6)$$

Here, Δr is the thickness of the surface (loose) particle layer and R is the radius of the dense particle core; obviously, $2(R + \Delta r) = d$. Knowing the value in the left-hand side of Eq. (6) from the A_{S2}/A_{S1} data (inset in Fig. 7), as well as the blocking temperature distribution functions $f(T_B)$ (Fig. 4) and the size distribution functions $f(d)$ (Fig. 5), we can obtain the R and Δr values as functions of the total particle diameter d . It was found that, for a particle with the

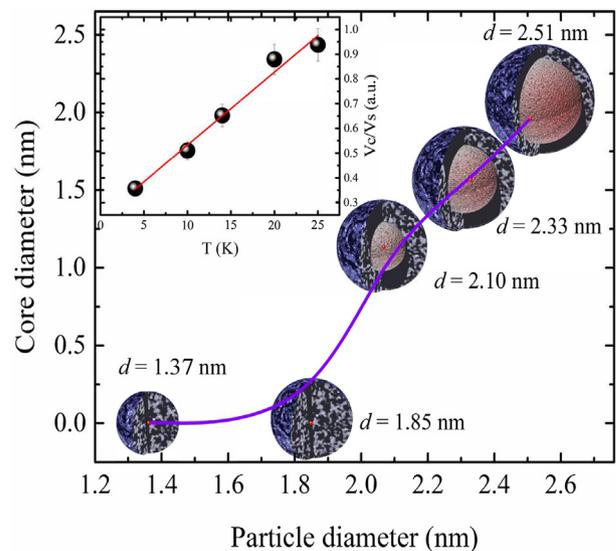


Fig. 7. Dependence of the ferrihydrite particle core size on the particle diameter. Inset: temperature dependence of the core-to-surface volume ratio according to the Mössbauer spectroscopy data.

maximum size ($d \approx 3$ nm), the surface layer thickness Δr is smaller than 0.3 nm. As the particle size decreases, the ratio between the volumes of the dense core and the porous surface decreases. For example, in a particle with a diameter of $d \approx 2.2$ nm, we have $\Delta r \approx 0.45$ nm at $2R \approx 1.3$ nm, whereas, in the case of particles with sizes d smaller than 2 nm, we can no longer speak about the dense core and, in fact, over the entire particle volume, the states of iron are similar to those of the porous surface. The aforesaid is illustrated in Fig. 7, which shows the dependence of the size of the dense core of ferrihydrite particles (in our designations, $2R$) on the particle diameter d . In addition, the figure shows the proposed model representation of ferrihydrite nanoparticles of different sizes with the distinguished regions of different densities (different states of iron atoms, according to the Mössbauer spectroscopy data).

It was obtained in Ref. [19] by analyzing the thermodynamic stability of ferrihydrite particles that, at a particle size of smaller than 1.5 nm, the amount of Gibbs free energy spent to the formation of the surface ($\sim R^2$) is larger than the gain resulting from the nucleation ($\sim R^3$). In other words, such particles are kind of surface aggregates with the branched surface and without the bulk crystalline core. This is consistent with our results, according to which a crystalline core starts forming only at a particle size of larger than 2 nm (Fig. 7). In this case, both the dense and porous phases exhibit the stable thermochemical properties [19]. Thus, ferrihydrite is a heterogeneous formation in which, after exceeding a certain particle size, a dense crystalline core is formed. Such a particle growth model determines the magnetic properties of the investigated material. The rise of the particle surface contribution should result in strong surface magnetism. Therefore, the magnetic ordering within the particle may proceed from the competition between the core and surface magnetic properties. Furthermore, interactions between surface states occur [75,76]. But in the case of biogenic ferrihydrite, we believe that the surface interparticle interactions are negligible due to the polysaccharide bacterial coating. Also, it is seen from the Mössbauer spectra form.

Reducing the particle size, we enhance the surface impact in the magnetism. According to the average particle size, we deal with the single-domain particles. Because of such ultra-small size, the particle diameter tends to the magnetic coherence length, and the magnetic ordering temperature can be much lower in comparison with bulk samples. It was established, for example, in the investigation of NiO nanoparticles [77,78]. The authors demonstrated a crucial decrease of the magnetic ordering temperature from 523 K (bulk sample) to 56 K (nanosized sample). In different oxides, magnetic correlation length may vary from the value of few nanometers ($\gamma - \text{Fe}_2\text{O}_3$, MnO, Fe_2O_3) [79–81] up to tens of nanometers in Co-doped $\gamma - \text{Fe}_2\text{O}_3$ [82]. Unfortunately, our experimental techniques do not allow us to determine the magnetic characteristic length. However, a comparison of the data obtained by M. S. Seehra and co-authors [22] reveals that magnetic coherence length is compared with the particle size. Therefore, we could state that at least magnetic coherence length does not exceed the polysaccharide coating of the biogenic ferrihydrite nanoparticles.

4. Conclusions

The biogenic ferrihydrite sample were formed by the activity of the *Klebsiella oxytoca* bacteria. Using the TEM investigations, the average size of the nanoparticles was found to be 2 nm. The Mössbauer spectra and the magnetic properties of the obtained sample were measured in the temperature range of 4–300 K. The analysis of the blocking/unlocking of the particle magnetic moment showed the dominating role of the surface magnetic anisotropy contribution to these processes and allowed us to calculate the bulk and surface magnetic anisotropy constants K_V and K_S ($K_V = 1.2 \cdot$

10^5 erg/cm³ and $K_S = 0.1$ erg/cm²). Using these data, the particle size distribution curve was calculated and appeared to be in good agreement with the TEM data. It was experimentally shown that the magnetic properties of ferrihydrite nanoparticles are caused by the formation of core-shell systems. In the first approximation, these states correspond to the surface (loose) layer and the well-crystallized dense core of a particle. According to the Mössbauer spectroscopy data, such a model is valid for particles larger than 2 nm. This size is critical for the formation of a well-crystallized core in a ferrihydrite particle, whereas smaller particles consist entirely of a material with a lower density. Thus, the contribution of the surface to the magnetic properties of ferrihydrite appears to be variable and depends on both the total particle size and the ratio between the surface fraction and the particle volume.

CRedit authorship contribution statement

Yu.V. Knyazev: Writing - original draft, Methodology, Data curation, Investigation, Conceptualization, Project administration, Visualization, Writing - review & editing. **D.A. Balaev:** Investigation, Data curation, Funding acquisition, Visualization, Writing - original draft, Methodology, Software, Writing - review & editing. **S.V. Stolyar:** Investigation, Funding acquisition. **O.A. Bayukov:** Validation, Software, Formal analysis, Investigation, Software. **R.N. Yaroslavtsev:** Investigation, Visualization. **V.P. Ladygina:** Investigation, Software. **D.A. Velikanov:** Investigation, Software. **R.S. Iskhakov:** Supervision.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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