Materials Advances

Accepted Manuscript

This article can be cited before page numbers have been issued, to do this please use: Y. Ma, S. Li, J. Wei, B. Quan, W. Liao, M. Molokeev, M. Cheng, X. Chen, Z. Zhou and M. Xia, *Mater. Adv.*, 2023, DOI: 10.1039/D3MA00654A.



This is an Accepted Manuscript, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

View Article Online

View Journal

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. We will replace this Accepted Manuscript with the edited and formatted Advance Article as soon as it is available.

You can find more information about Accepted Manuscripts in the Information for Authors.

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard <u>Terms & Conditions</u> and the <u>Ethical guidelines</u> still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this Accepted Manuscript or any consequences arising from the use of any information it contains.



rsc.li/materials-advances

This article is licensed under a Creative Commons Attribution-NonCommercial 3.0 Unported Licence

Open Access Article. Published on 19 October 2023. Downloaded on 10/30/2023 3:18:38 AM

ARTICLE

Received 00th January 20xx, Accepted 00th January 20xx

DOI: 10.1039/x0xx00000x

Spectroscopically enhanced far-red phosphor Li₂Mg₃TiO₆: Cr³⁺ and its application prospects to cold resistance of rice

Yibiao Ma^{a,b}, Siying Li^{a,b}, Jiaqi Wei^{a,b}, Weifang Liao^{a,b}, Beibei Quan^{a,b}, Maxim S. Molokeev^{a,d}, Ming Cheng^c, Xiaoyan Chen^c, Zhi Zhou^{*,a,b} and Mao Xia^{*,a,b}

Chemical unit co-substitution is a very effective strategy to improve the properties of phosphors. Due to the mismatch of the radii between substituted ions, unexpected properties are usually produced. Such properties are of profound significance to expand the research field. In this study, $Mg^{2+}-Ti^{4+}$ in Li₂ Mg_3TiO_6 : Cr³⁺ was replaced by Al³⁺-Al³⁺ and Ga³⁺-Ga³⁺, while charge balance was maintained. Ion substitution changed the crystal field environment of activator ion, which increased the luminescence intensity by 180% and 184% respectively, accompanied by a slight decrease in thermal stability. In addition, the quantum efficiency was increased from 35.1% to 73.1%. The electroluminescence spectrum of the encapsulated pc-LED was examined, and the overlap with the absorption profile of the phytochrome P_{fr} was 61%. In order to verify the application prospect of far-red phosphor. A 15-day rice growth experiment was set up to detect surface traits, soluble sugars, soluble proteins, and the expression of OsphyA and OsCBF3 genes. It was demonstrated that rice under far-red light irradiation had significant resistance enhancement.

1. Introduction

Since human society entered the 21st century, the process of social industrialization has been accelerating, leading to a deterioration of the environment and a decrease in agricultural production¹⁻³. Crops grown outdoors are highly susceptible to extreme weather, which can make it difficult to get high-quality food. Plant factory, as efficient agricultural facilities, can handle these issues well^{4, 5}. The most important component in a plant factory is the light source. Light sources are indispensable for plant growth and affect the whole period of the plant's growth. Light acts on plants through photoallergens⁶. Typically, plants absorb both blue and red light, with red light being divided into red light and far red light. This corresponds to the response of different biological pigments in plants to light: chlorophyll A absorbs red light in the 420 nm to 663 nm, chlorophyll B absorbs blue light in the 460 nm to 645 nm range, the phytochrome P_r absorbs red light with a peak at 660 nm and phytochrome P_{fr} absorbs far-red light with a peak at 730 nm^{7, 8}. In the field of plant factories, the light source used for plant lighting is usually pc-LED. This complementary light method has the advantage of being small and energy-efficient, and can accurately correspond to the spectral absorption of the plant. Zhong and Zhou et al. investigated the method of blue LED chip and red phosphor for plant lighting with remarkable results, which marks the broad application prospects of pc-LED in the plant field⁹. In addition to blue and red light, far-red light can affect plant growth and development. Generally, far-red light has two effects on plants: shading and flowering induction¹⁰. When exposed to far-red light, the plants appear to be blocked by other leaves, which causes them to grow taller for more sunlight. From this, the plant height can be tuned by adjusting the red/far-red light ratio. liu et al¹¹. reported a novel phosphor Ga₂O₃: Cr³⁺ that perfectly matches the absorption of phytochrome P_{fr}, and two kinds of plants "Aglaonema" and "P. amboinicus" were selected for application study. Gai and Xia et al. also prepared Mg₂SnO₄:Cr³⁺ phosphor and the experiment of tomato quality was carried out¹².

Mn⁴⁺ is considered the most typical activator of phosphors used in plant lighting, and can emit red or far-red light, for example: LaAlO₃: Mn⁴⁺(centered at 726 nm)¹³, SrLaAlO₄: Mn⁴⁺(centered at 730 nm)¹⁴, CaYAlO₄: Mn⁴⁺(centered at 713 nm) ¹⁵. With the development of science and technology, it was gradually discovered that the non-rare earth element Cr³⁺ also has excellent properties. Specifically, compared to Mn⁴⁺, the luminescence of Cr³⁺ is more susceptible to the crystal field strength (CFS), under which Cr³⁺ can produce a tunable spectrum from the far-red to the near-infrared. The excitation range is from 200 to 650 nm. Most studies have shown that the boundary Dq/B value between strong and weak crystal fields is 2.3. In a strong crystal field, the electron makes a ${}^{2}E \rightarrow {}^{4}A_{2}$ transition, emitting far-red light. When in the weak crystal field, ${}^{4}T_{2} \rightarrow {}^{4}A_{2}$ transitions, emitting near infrared light. In the intermediate crystal field, ${}^{4}T_{2} \rightarrow {}^{4}A_{2}$ transitions and ${}^{2}E \rightarrow {}^{4}A_{2}$ transitions and emit near infrared and far-red light^{16, 17}. Due to this property, Cr³⁺ excited phosphors can be widely used in the field of near-infrared luminescence. For plant illumination, the performance of Cr³⁺ in farred illumination is utilized. For example, the emission peak of phosphors such as Na₃AIF₆: Cr³⁺, K₃AIF₆: Cr³⁺, Rb₂NaAIF₆: Cr³⁺, KMgF₃:

^{a.} School of Chemistry and Materials Science, Hunan Agricultural University, Changsha 410128, PR China.

^{b.} Hunan Optical Agriculture Engineering Technology Research Center, Changsha 410128, PR China.

^{c.} Dongguan Ledstar Optoelectronics Technology Co., Ltd.

^{d.} World-Class Research Center "Advanced Digital Technologies", University of Tyumen, Tyumen 625003, Russia.

Electronic Supplementary Information (ESI) available: [details of any supplementary information available should be included here]. See DOI: 10.1039/x0xx00000x

This article is licensed under a Creative Commons Attribution-NonCommercial 3.0 Unported Licence Open Access Article. Published on 19 October 2023. Downloaded on 10/30/2023 3:18:38 AM

ARTICLE

Cr³⁺, is around 730 nm¹⁸⁻²¹. However, these phosphors all contain fluoride, and hydrofluoric acid is used in the preparation process, which is extremely unfriendly to agricultural ecology and the social environment. Therefore, the current research focuses on oxide phosphors, and also have excellent luminous properties. Garnet phosphor Gd₃Al₂Ga₃O₁₂: Cr³⁺ (IQE=97.3%, EQE=70.1%)²², and has excellent thermal stability. Other garnet-structured phosphors such as Gd_{2.4}Lu_{0.6}Ga₄AlO₁₂: Cr^{3+ 23}, Gd₃Sc_{1.5}Al_{0.5}Ga₃O₁₂: Cr^{3+ 24} also have good photometric properties.

Chemical unit co-substitution can tune the emission spectrum in an efficient way by simultaneously changing the two structural units to adjust the coordination environment of the activator ions. Zhong et al. improved the quantum efficiency of $Ca_{14}Ga_{10}Zn_6O_{35}$:Mn⁴⁺ from 38.0% to 50.9% by unit co-substitution strategy²⁵. Zhang et al. substituted W⁶⁺-Ga³⁺, W⁶⁺-Al³⁺ for Ti⁴⁺-Sb⁵⁺ and increased the luminescence intensity of LaTiSbO₆: Mn⁴⁺ by 204% and 182%, respectively²⁶. Li₂Mg₃TiO₆: Cr³⁺ has perfect octahedral site points, which can provide a coordination environment for Cr³⁺ ions. In addition, it is exciting to adjust the emission peak shift of this phosphor beyond 200 nm by the concentration of Cr³⁺²⁷ (from 720 nm to 920 nm).

In this work, Li₂Mg₃TiO₆:Cr³⁺ phosphor was successfully prepared. In previous studies, it has been known that the phosphor emission peak is affected by the Cr³⁺-Cr³⁺ substitution of Mg²⁺-Ti⁴⁺²⁷, so we have chosen the appropriate Cr³⁺ concentration for better optimization. By replacing the Mg²⁺-Ti⁴⁺ sites with Al³⁺-Al³⁺ and Ga³⁺-Ga³⁺ ions, the far-red emission of the material is enhanced with a slight decrease in thermal stability. This is a valid and exciting result. A lot of testing and analysis was done. Finally, LED devices packaged with blue-light chips and the experiment on the improvement of cold resistance of rice by far red light was added. This indicates that asprepared phosphor has a wide range of potential applications in the field of plant lighting.

2. Experimental sections

2.1 Samples synthesis

A series of Li₂Mg₃TiO₆:Cr³⁺ phosphor were prepared by high temperature solid phase method. Li₂CO₃ (99.99%), MgO (99.99%), TiO₂(99.99%), and Cr₂O₃ (99.99%), Al₂O₃(99.99%), Ga₂O₃(99.99%) was used as raw material, weighed by stoichiometric ratio, transferred to an agate mortar for full reaction for 30min, and then the mixture was filled into a corundum crucible for firing in two steps: First, it was fired for 4h in the air atmosphere of the box furnace at 800°C. After cooling, it was re-ground and then fired for 6 h at 1280°C. All fired samples are ground into powder for subsequent testing.

2.2 Samples characterizations

The X-ray diffraction (XRD) patterns of the powders were obtained by using a diffractometer (Bruker, Germany) with the Cu-K α radiation and test range of 10-80°. The fluorescence spectra (PL and PLE), thermal resistance and quantum yield were detected by a fluorescence spectrophotometer (Edinburgh FLS1000, UK). The UVvis diffuse reflectance (DR) spectra was obtained by a spectrophotometer (SHIMADZU UV-2600i, Japan). The electroluminescence (EL) spectra and electro-optical conversion efficiency of the LED devices were recorded by a measurement system (ATA-500, Everfine, China).

2.3 PC-LED fabrication

Using A mixture of LMTGO: Cr^{3+} and epoxy resin (A:B=1:10), dot coating on the blue light chip. Transfer it to the oven at 120°C for 12 hours to fully cure it and conduct subsequent tests.

2.4 Experiment on rice growth variation

2.4.1 Crops grown

The experiment was conducted in the optical plant laboratory of Hunan Agricultural University, with the rice variety XiangZaoxian24, at a temperature of 25±5°C and humidity of 50±10%. The planting area was 60×120×200 (cm).

2.4.2 Planting conditions

Disinfection with a solution of sodium hypochlorite at a concentration of 5% for 20 minutes. The seeds were repeatedly rinsed with tap water and soaked in more than 1 L of water for 24 h. The water was changed every 12 h. The sterilized soaked rice seeds were placed in a 37 °C incubator to germinate for 36 h. Rice with a root length of about half a grain was selected and sown on a seedling tray with a size of 30×60×10 (cm) and a total of 169 holes, with 4 seeds per hole, using nutrient-free vermiculite as the substrate. The whole nutrient solution was used for cultivation and planting, and the nutrient solution was changed every 3 days. After 15 days of planting, some plants were randomly selected for follow-up testing. The plants were then subjected to white light at a low temperature of 4°C and 400 ± 10ummol/s light intensity for one day to simulate the phenomenon of spring cold, and then a portion of the sample was stored in an ultra-low temperature refrigerator at -80 °C for subsequent testing.

2.4.3 Rice trait testing

Plant height and stem base width are measured by tape rulers and vernier calipers respectively. Above-ground and below-ground parts of rice seedlings planted under two light conditions for 15 d were taken separately, rinsed with distilled water, and then the surface water was drained with filter paper and the fresh weight was weighed, after which they were transferred to an oven and dried at 105°C to a constant weight and weighed as dry weight. Weighing 0.2 g fresh sample, grinding it well, setting the speed at 3000r/min, centrifugation for 10 min, taking 1.0 mL supernatant, adding 5 mL of 100 mg/L Coomasil bright blue, shaking well, and placing it for 2 min. Absorbance was measured with 595 nm enzyme spectrometer, and soluble protein content was calculated. The expressions of OsphyA and OsCBF3 genes were analyzed by real-time fluorescence quantitative RT-PCR with a reaction volume of 10 μ L. Includes cDNA 2 μL, Hieff[®]qPCR SYBR GreenMasterMix 5 μL, forward primer 0.2 μL, reverse primer 0.2 µL, and ddH2O 2.6 µL.

3. Results and discussion

3.1 Phase Identification and Crystal Structure

This article is licensed under a Creative Commons Attribution-NonCommercial 3.0 Unported Licence.

) BΥ-NC

3

 $R_{
m wp}$,%

 $R_{\rm p}$,%

 χ^2

Open Access Article. Published on 19 October 2023. Downloaded on 10/30/2023 3:18:38 AM.

ARTICLE



Fig.1(a) XRD spectrum of $Li_2Mg_3TiO_6$:xCr³⁺ compared with $Li_2Mg_3SnO_6$ standard card; (b) Refinement of $Li_2Mg_3TiO_6$; (c) Crystal structure diagram and chemical unit co-substitution schematic model of $Li_2Mg_3TiO_6$.

 $R_{\rm B},\%$

Compound	Li ₂ Mg ₃ TiO ₆	
Sp. Gr.	Fm-3m	
a (Å)	4.17823(3)	
V (ų)	72.9421(17)	
2ϑ-interval, º	20-120	

8.47

6.49

1.32

Table 1. Main parameters of processing and refinement of the

Table 2. Fractional atomic coordinates and isotropic displacement parameters (Å²) of $Li_2Mg_3TiO_6$

0.82

		2 00 0				
Atom	x	у	Ζ	B _{iso}	Occ.	
Li	0	0	0	0.6(3)	1/3	
Ti	0	0	0	0.6(3)	1/6	
Mg	0	0	0	0.6(3)	1/2	
0	0.5	0.5	0.5	1.0(3)	1	

ARTICLE

Fig.1a shows the XRD patterns of Li₂Mg₃TiO₆ doped with different Cr³⁺ concentrations. Due to the absence of JCPDS of Li₂Mg₃TiO₆, Li₂Mg₃SnO₆ (JCPDS 39-0392) with similar crystal structure was used as the standard card for comparison, and professional software was used for model fitting. As shown in fig.1b, the results show that the crystal structure is fully consistent with Li2Mg3TiO6 when the ratio of Li, Mg and Ti is constrained to 2:3:1 for refinement. Detailed parameters are listed in Table 1-2. It can be observed from the illustration of fig.1a that the diffraction peak near 43° gradually moves to a higher angle, which is caused by the gradual increase of Cr³⁺ concentration. Cr³⁺ (0.615 Å, CN = 6) and Ti⁴⁺ (0.605 Å, CN = 6) have very similar ionic radii, while Mg²⁺ (0.72 Å, CN = 6) has larger radii. This peak shift proves that Cr³⁺-Cr³⁺ replaces Mg²⁺-Ti⁴⁺ in LMTO. Fig.1c shows the crystal structure model of the LMTO. Li⁺, Mg²⁺ and Ti⁴⁺ occupy a common octahedral position point with a ratio of 2:3:1, connects oxygen atoms at the same bond length. The XRD diffraction patterns obtained after the substitution of Al³⁺-Al³⁺ and Ga³⁺-Ga³⁺ are archived in fig. S1 of the Supplement information, and the measured diffraction peaks are consistent with the standard cards of Li2Mg3SnO6.

3.2 Spectroscopic properties.

High concentrations of Cr3+ doping will cause the LMTO emission to shift into the NIR region. When the concentration of Cr³⁺ is limited to less than 1 %, the emission peak does not shift and the intensity gradually increases over a concentration range of 0.001 - 0.009. When the concentration of Cr³⁺ gradually increases, it is shown in Fig.2a and b. Three excitation bands exist from 250 nm to 600 nm, located at 330 nm, 450 nm and 590 nm respectively. The optimum excitation band is from 400 nm to 500 nm. The emission spectra range from 600 nm to 850 nm with a peak at 707 nm. The first excitation peak in the UV region is attributed to the charge transfer band (CTB) of Cr-O, and the second and third peaks are from the spinallowed ${}^{4}A_{2} \rightarrow {}^{4}T_{1}$ (⁴F) and ${}^{4}A_{2} \rightarrow {}^{4}T_{2}$ (⁴F) transitions of Cr³⁺, respectively (Fig.2c). It is worth noting that Cr³⁺ emission generally has both an Rline and a broadband, attributed to the ${}^{2}E \rightarrow {}^{4}A_{2}$ and ${}^{4}T_{2} \rightarrow {}^{4}A_{2}$ transitions of Cr3+, respectively28. The generation of these two transitions depends on the CFS around Cr3+. In strong crystal field systems, R-line emission dominates, while in weak crystal field systems, broadband emission dominates, while in some phosphors, both R-line and broadband emission transitions are present. The CFS is judged by Dq/B = 2.3 as a cut-off value²⁹. When Cr^{3+} occupies the octahedral site, the CFS can be calculated according to the following equation^{30, 31}:

$$Dq = \frac{E({}^{4}A_{2} \rightarrow {}^{4}T_{2})}{10}$$
 (1)

$$\frac{Dq}{B} = \frac{15(x-8)}{(x^2 - 10x)}$$
(2)

$$x = \frac{E^{(4} A_2 \to {}^4 T_1) - E^{(4} A_2 \to {}^4 T_2)}{Dq}$$
(3)

Where E $({}^{4}A_{2} \rightarrow {}^{4}T_{1})$ and E $({}^{4}A_{2} \rightarrow {}^{4}T_{2})$ represent the energy difference between ${}^{4}A_{2}$ levels to ${}^{4}T_{1}$ and ${}^{4}T_{2}$ levels, which can be determined by the position of excitation peaks. Determined as 22222 cm⁻¹ and 16949 cm⁻¹. The calculated Dq/B value for LMTO: Cr³⁺ is 3.57,

this represents a strong crystal field and consistent with the previously reported far-red emission of Cr^{3+} in a strong crystal field.

x=0.001 x=0.002 x=0.004 x=0.006 x=0.008 x=0.00 x=0.00 x=0.00 x=0.00 Li2Mg3TiO6: xCr3 (a.u.) (a.u.) x=0.004 Li₂Mg₃TiO₆: =450 nm Intensity Intensity 800 Wavelength (nm) Wavelength (nm) d с m=707 nm Experimental data Fitting curve 1 Fitting line Fitting curve 2 Fitting curve 3 R²=0.984 Intensity (a.u.) 5.5 g(I|x) $R^2 = 0.948$ 5.3 Slope= -0.601 5.2 -2.8 -2.6 -2.4 -2.2 -2.0 Wavelength (nm) lg(x)

Fig. 2(a-b) Excitation and emission spectra of $Li_2Mg_3TiO_6$ for different Cr^{3+} concentrations, (c) Gaussian fitted $Li_2Mg_3TiO_6$: 0.006 Cr^{3+} excitation profile (d) Relationship between log (x) and log (1/x).

The illustration of Fig.2b shows the dotted line plot of Cr^{3+} concentration addition, when x = 0.006, the luminous intensity is the strongest, then the luminous intensity decreases due to concentration quenching, the critical distance (R_c) can be used to explain the mechanism of this phenomenon^{32,33}:

$$R_C \approx 2 \left(\frac{3\nu}{4\pi x_c N}\right)^{\bar{3}} \tag{4}$$

Where *N*, X_c and *V* represent the possible sites occupied by Cr³⁺, the activator ion concentration and cell volume respectively. The calculated R_c is 28.53, the value is greater than 5 Å, so the mechanism of energy transfer is a multipole-multipole interaction. The specific interactions can be calculated from the following equations^{33, 34}:

$$\frac{d}{dx} = K \left[1 + \beta(x)^{\theta/3} \right]^{-1}$$
(5)

Where x represents the concentration of Cr^{3+} ions and *I* represents the luminous intensity of Cr^{3+} ions. Under the same test conditions, both k and β are constants. The slope shown in Fig.2d is -0.6, so the value of x is 1.8, which is close to 3. Therefore, exchange interactions are the mechanism of concentration quenching in LMTO: Cr^{3+} .

This article is licensed under a Creative Commons Attribution-NonCommercial 3.0 Unported Licence

Open Access Article. Published on 19 October 2023. Downloaded on 10/30/2023 3:18:38 AM





Fig.3(a-b) Excitation and emission patterns of LMTO: Cr³⁺ after Al³⁺-Al³⁺ substitution; (c-d) Excitation and emission patterns of LMTO: Cr³⁺ after Ga³⁺-Ga³⁺ substitution.

Fig.3 shows the excitation and emission spectra following the addition of the Al³⁺-Al³⁺ and Ga³⁺-Ga³⁺ ion pairs. When the ion pair replaces the Mg²⁺-Ti⁴⁺ ion pair in small amounts, the luminescence of the material appears significantly enhanced and the spectrum is unchanged. As the amount of substitution continues to increase, the luminescence appears to diminish, with the optimum concentrations of 0.015 and 0.01 for the two substitution strategies, enhancing the luminescence to 180% and 184%, respectively.

3.3 Reflectance spectra and optical bandgap

Fig.4a shows the diffuse reflectance spectra of LMTO host, LMTO: Cr³⁺, LMTAO: Cr³⁺ and LMTGO: Cr³⁺, and the excitation spectra at 707 nm were chosen to correspond to them. The test interval for diffuse reflectance is from 800 nm to 200 nm, with the LMTO host appearing as an almost flat line in the 800 nm - 400 nm range, with a sharp dip after 400 nm. Three valleys of varying intensity appear in the region around 600 nm, 500-400 nm, and 400-300 nm, which correspond to the ${}^{4}A_{2} \rightarrow {}^{4}T_{2}$, ${}^{4}A_{2} \rightarrow {}^{4}T_{1}$ and Cr-O charge transfer bands, consistent with the excitation spectrum. In addition, it can be observed that the reflectance of the chemical unit co-substituted material becomes lower, demonstrating stronger absorption, LMGT: Cr³⁺ absorption is stronger, again corresponding to the results of the fluorescence spectrum. The band gap (E_{gap}) was calculated by the following equation³⁵⁻³⁷:

$$F(R_{\infty}) = (1 - R_{\infty})^2 / (2R_{\infty})$$
(6)

$$(\alpha h v)^n = (h v - Eg) \tag{7}$$

 R_{∞} represents the reflectance, α is the absorption parameter, hv is a proportional constant, *n* is defined as 1/2, and E_a is the optical band gap. The calculated optical band gaps for the four materials are 3.93 eV, 3.92 eV, 3.88 eV and 3.87 eV (Fig.4b-e). There is a slight tendency for the band gaps to become smaller, making electron-hole recombination easier.

DOI: 10.1039/D3MA00654A

ARTICLE



Fig.4 Diffuse reflectance spectra and optical band gaps of LMTO: host, LMTO: Cr³⁺, LMTAO: Cr³⁺ and LMTGO: Cr³⁺.

3.5 Temperature-Dependent Luminescence and Quantum Efficiency

Elevated internal temperature of the phosphor induces heightened non-radiative transitions and consequent energy dissipation. More than that, after phosphor and blue light chip combination, large current makes pc-LED high power operation, resulting in a lot of heat. In plant lighting, it is difficult to use high power radiator discharge heat, accumulated heat will seriously affect the performance of phosphor, so it is very necessary to evaluate the resistance of phosphor to high temperature environment. The luminescence properties of phosphors from 298K to 423K are tested. As the temperature increases (Fig. 5a-c), the luminosity intensity of LMTO: Cr3+ decreases significantly. When it reaches 423K, the luminous intensity maintains 57.9 % of that at room temperature. The two materials after co-substitution of chemical units are shown in Fig. 5d-i. Similarly, with the increase of temperature, the luminescence properties of the materials decrease, which may be due to the structural stiffness of the materials decreases after the cosubstitution of chemical units, leading to the increase of energy lost by non-radiative transitions. In order to verify this conjecture, the thermal quenching activation energy of phosphor was calculated by the following Arrhenius formula³⁸:

$$I(T) = \frac{I_0}{1 + Aexp\left(\frac{-E_a}{kT}\right)} \tag{8}$$

 I_0 and I (T) represent the emission intensity at room temperature and at temperature T respectively, k is Boltzmann's constant (k = 8.62 \times 10⁻⁵ eV), and A is a constant.



Fig. 5(a-c) Temperature dependent luminescence spectra of LMTO: Cr^{3+} from 298K to 423K; (d-f) Temperature-dependent emission spectra of LMTAO: Cr^{3+} ; (g-i) Temperature-dependent emission spectra of LMTGO: Cr^{3+} .

The mechanism can also be better understood by plotting configuration coordinates (Fig. S2): when receiving energy, electrons transition from the ground state to the excited state, then some return to the ground state by radiative transition, and others return to the ground state by non-radiative transition at the intersection of the $^{4}A_{2}$ and $^{4}T_{2}$ energy levels, resulting in a loss of energy. The vertical distance from the intersection of $^{4}A_{2}$ and $^{4}T_{2}$ to the bottom of the $^{4}A_{2}$ level is the thermal quenching activation energy. The calculated E_a values of LMTO: Cr³⁺, LMTAO: Cr³⁺ and LMTGO: Cr³⁺ are 0.283 eV, 0.246 eV and 0.245 eV (Fig. 6), respectively. The decrease of thermal quenching activation energy shows that the thermal quenching resistance of phosphor decreases, which is consistent with the experimental results.



Fig. 6 Thermal quenching activation energy of LMTO: Cr^{3+} , LMTAO: Cr^{3+} and LMTGO: Cr^{3+} .

Quantum efficiency is an important parameter to evaluate the performance of phosphors. After testing and calculation, the

DOI: 10.1039/D3MA00654A Journal Name

Page 6 of 9

quantum yield of LMTO: Cr^{3+} and LMTGO: Cr^{3+} is increased from 35.1% to 73.1% (Fig.7) . The quantum yield can be derived from the following formula 39 :

$$\eta = \frac{\int Ls}{\int E_R - \int E_S}$$
(9)

In the formula, η represents the internal quantum efficiency, L_s represents the sample emission spectrum, E_R represents the excitation spectrum of BaSO₄, and E_s represents the excitation spectrum of the sample.



Fig. 7 Quantum efficiency of LMTO: Cr³⁺ and LMTGO: Cr³⁺.

3.6 Fluorescence lifetime and Electroluminescent properties Characters

The fluorescence lifetime of phosphors is shown in Fig. 8a, which is obtained by fitting the following double exponential model^{40, 41}:

$$I(t) = I_0 + Mexp\left(-\frac{t}{\tau_1}\right) + Nexp\left(-\frac{t}{\tau_2}\right)$$
(10)

$$\tau_{ave} = \frac{(c_1 \tau_1^2 + c_2 \tau_2^2)}{(c_1 \tau_1 + c_2 \tau_2)} \tag{11}$$

Herein, τ_{ave} represent the average lifetime. The fluorescence lifetime of LMTO: Cr³⁺, LMTAO: Cr³⁺ and LMTGO: Cr³⁺ were 2.237 ms, 2.314 ms and 2.470 ms, respectively.

LMTGO: Cr³⁺ phosphor and 450 nm blue light chip were used to package LED devices, and the measured electroluminescent spectrum was shown in Fig. 8b. The obtained far-red emission can be well matched with the P_{fr}, The overlap after normalization is 61%, indicating that LMTGO: Cr³⁺ phosphors have good application potential in agriculture.

Open Access Article. Published on 19 October 2023. Downloaded on 10/30/2023 3:18:38 AM.

View Article Online DOI: 10.1039/D3MA00654A ARTICLE



Fig. 8 (a)The fluorescence lifetime of LMTO: Cr^{3+} , LMTAO: Cr^{3+} and LMTGO: Cr^{3+} (b)Comparison between the electroluminescence spectra of LMTGO: Cr^{3+} and the absorption spectra of plant pigments, (c-d) Rice growing under white light and white light + far red light.

3.6 Experiment on cold resistance of rice

Rice originated in the tropics and is widely cultivated around the world, with more than half of the world's population relying on rice as their primary food. After rice breeding, even if transplanted to the southern region, it is often faced with a sudden temperature drop below 10 °C, and even snow and ice weather. This extreme climate can severely affect the physiological state of the rice, negatively affecting its overall metabolism and grain yield. In this work, rice growing under white light was used as a control, and far-red light radiation was added to the experimental group to explore the effect on cold resistance of rice, as shown in Fig. 8c-d.



Fig. 9 Apparent data of rice under white light and white light plus far-red light. Vertical bars present the mean value. Capped bars above represent the standard error. Duncan's multiple range test (P < 0.05) was used to test for differences between averages.

Page 8 of 9

ARTICLE



Fig. 10 (a)Soluble total sugar content, (b)soluble protein content, (c)relative expression of phyA gene, and (d)relative expression of CBF gene in rice under white light and white light plus far-red light. Vertical bars present the mean value. Capped bars above represent the standard error. Duncan's multiple range test (P < 0.05) was used to test for differences between averages.

Two groups of rice growing under white light irradiation and far red light + white light irradiation were selected as research objects. (Fig.8c-d) After 15 days of growth, Phenotypic images and trait data of rice were obtained, scale bars=10 cm (c and d). The addition of some far-red light irradiation under white light increased the stem base width and the dry and fresh weights of the above-ground parts of rice (Fig. 9). The plant height decreased by 14.94%, the stem base width increased by 6.29%, and the whole plant fresh weight increased by 1.14%, which finally led to an increase of 16.6% in the sturdy seedling index of rice, which means that rice will be stronger. More importantly, plant traits are generally influenced by gene expression, for example, increased expression of the photosensitive pigments phyA and CBF genes can improve the robustness index of rice by affecting the ABA/GA ratio. To investigate the mechanism of the effect of far-red light on cold tolerance in rice, some representative changes in sugar, protein and gene expression were studied. As shown in Fig. 10, the soluble sugars of rice under far-red light irradiation increased by 26.21% and soluble proteins increased by 1.08%. And for the gene expression, the expression of phyA increased by 277.03% and the expression of CBF3 increased by 425.64%.

Conclusions

In this work, Li₂Mg₃TiO₆: Cr³⁺ phosphor was successfully prepared, and XRD patterns showed that the material was successfully synthesized and did not contain other impurities. The photoexcitation spectra show that the material has an optimal excitation band of 400-500 nm and an emission band of 600-850 nm (peak at 707 nm). By replacing Mg²⁺-Ti⁴⁺ with Ga³⁺-Ga³⁺ and Al³⁺-Al³⁺ ions, the luminosity of the Li₂Mg₃TiO₆: Cr³⁺ phosphors was increased by 180% and 184%, respectively, with a slight thermal quench (from 57.9% to 51.1% and 50.3%, respectively, at 423K). The variation in luminosity properties is explained by the thermal activation energy and the optical band gap. Even more exciting is that the quantum efficiency of the phosphor has been improved by 2.08 times. After conducting experiments on rice growth, it was found that rice irradiated with far-red light had better resistance to cold, which could be analyzed and explained at the epigenetic and genetic level: Plant height and stem width decreased, seedling strength index increased by 16.6%, soluble total sugar increased by 26.21%, soluble protein increased by 1.08%, OsphyB expression increased by 277.03%, CBF3 expression increased by 425.64%. Such an exciting result is a sign of the excellent application of this job in the field of biological lighting.

Acknowledgements

This work was supported by the National Natural Science Foundation of China (Grant No. 51974123), the Key R & D Projects in Hunan Province (2021SK2047, 2022NK2044), the Natural Science Foundation of Hunan Province, China (Grant No. 2021JJ40261), Wangcheng Science and Technology Plan (KJ221017), the science and technology innovation Program of Hunan Province (2022WZ1022). The work was supported by the Ministry of Science and Higher Education of the Russian Federation as part of World-class Research Center program: "Advanced Digital Technologies", contract no. 075-15-2020-935.)

Author Contributions

Yibiao Ma: conceptualization, data curation and writing - original draft. Siying Li and Jiaqi Wei: conceptualization, data curation. Ming Cheng and Xiaoyan Chen: investigation and formal analysis. Beibei Quan and Weifang Liao: data curation. Maxim S. Molokeev: data curation and formal analysis. Zhi Zhou and Mao Xia: resources, writing - review & editing and supervision.

Conflicts of interest

There are no conflicts to declare.

Notes and references

- 1 G. Annadurai, B. Devakumar, H. Guo, R. Vijayakumar, B. Li, L. Sun, X. Huang, K. Wang; X.W. Sun, *RSC Adv.*, 2018, *8*, 23323-23331.
- 2 X. Huang; H. Guo, Dyes Pigments, 2018, 152, 36-42.
- 3 S. Zhang, S. Gai, X. Zhang, M. Xia, Z. Zhou, X. Cheng, M. Yao, M.S. Molokeev, Q. Feng, *Ceram. Int.*, 2022, 48, 3070-3080.
- 4 X.J. Kang, W.W. Yang, D.X. Ling, C.A.Y. Jia, W. Lu, *Mater. Res.* Bull., 2021, 140. 111301.
- 5 Z.Z. Lu, Y.B. Meng, H. Fan, J.Y. Lu, X.Y. Zhong, Y.J. Ou, H.R. Mo, L.Y. Zhou, *J. Lumin.*, 2021, 236. 118100.
- 6 N. Yeh, J.-P. Chung, *Renew. Sust. Energ. Rev.*, 2009, *13*, 2175-2180.
- 7 A. Sadanandom, E. Adam, B. Orosa, A. Viczian, C. Klose, C.J. Zhang, E.M. Josse, L. Kozma-Bognar, F. Nagy, P. Nat. Acad. Sci. Usa., 2015, 112, 11108-11113.
- 8 Z. Zhou, M. Xia, Y. Zhong, S. Gai, S. Huang, Y. Tian, X. Lu; N. Zhou, *J. Mater. Chem. C*, 2017, *5*, 8201-8210.
- 9 Z. Zhou, J. Zheng, R. Shi, N. Zhang, J. Chen, R. Zhang, H. Suo, E.M. Goldys, C. Guo, *Acs Appl. Mater. Inter.*, 2017, *9*, 6177-6185.
- 10 D.J. Sheerin, A. Hiltbrunner, *Plant Cell Environ.*, 2017, 40, 2509-2529.
- 11 M.-H. Fang, G.N.A. De Guzman, Z. Bao, N. Majewska, S. Mahlik, M. Grinberg, G. Leniec, S.M. Kaczmarek, C.-W. Yang, K.-M. Lu, H.-S. Sheu, S.-F. Hu, R.-S. Liu, *J. Mater. Chem. C*, 2020, *8*, 11013-11017.
- S. Gai, C. Zhou, L. Peng, M. Wu, P. Gao, L. Su, M.S. Molokeev, M. Xia, *Mater. Today Chem.*, 2022, *26*. 101107.
- 13 S.Y. Li, Q. Zhu, X.D. Sun, J.G. Li, *J. Mater. Chem. C*, 2021, *9*, 7163-7173.
- Y. Zhang, Y. Huang, M. Li, C. Liang, H. Zhu, Y. Zhong, N. Yang,
 Z. Zhou, M. Xia, J. Am. Ceram. Soc., 2020, 103, 4373-4383.
- 15 K. Sankarasubramanian, B. Devakumar, G. Annadurai, L. Sun, Y.J. Zeng; X. Huang, *RSC Adv.*, 2018, *8*, 30223-30229.
- 16 C. Zhou, L. Peng, Z.H. Kong, M.H. Wu, M.S. Molokeev, Z. Zhou, J. Wang; M. Xia, J. Mater. Chem. C, 2022, 10, 5829-5839.
- 17 S. Zhang, Y. Liu, J. Yin, X. Zhang, Y. Li, L. Su, Z. Zhou, M. Xia, Mater. Today Chem., 2022, 24. 100835.
- D. Yu, Y. Zhou, C. Ma, J.H. Melman, K.M. Baroudi, M. LaCapra, R.E. Riman, Acs Appl. Electron. Ma., 2019, 1, 2325-2333.
- C. Lee, Z. Bao, M.H. Fang, T. Lesniewski, S. Mahlik, M. Grinberg, G. Leniec, S.M. Kaczmarek, M.G. Brik, Y.T. Tsai, T.L. Tsai, R.S. Liu, *Inorg. Chem.*, 2020, *59*, 376-385.
- 20 D.X. Shi, Z.B. Liang, X. Zhang, Q. Zhou, Z.L. Wang, M.M. Wu, Y.Q. Ye, *J. Lumin.*, 2020, 226. 117491.
- 21 J.L. Pascual, J. Phys. Chem. C, 2019, 123, 27150-27164.
- D. Wu, H. Wu, Y. Xiao, X.L. Dong, Y. Wang, W.P. Zhou, Y.F. Liu,
 L.L. Zhang, J. Lumin., 2022, 244. 118750.
- 23 X. Zou, X. Wang, H. Zhang, Y. Kang, X. Yang, X. Zhang, M.S. Molokeev, B. Lei, *Chem. Eng. J.*, 2022, *428*. 132003.
- 24 E.T. Basore, H.J. Wu, W.G. Xiao, G.J. Zheng, X.F. Liu, J.R. Qiu, *Adv. Opt. Mater.*, 2021, *9*. 2001660.

- 25 Y. Zhong, S.J. Gai, M. Xia, S.M. Gu, Y.L. Zhang, X.B. Wu, J. Wang, N. Zhou, Z. Zhou, *Chem. Eng. J.*, 2019, 374, 381-391.
- 26 X.J. Zhang, Y.B. Ma, P.X. Gao, L.J. Su, Z.T. Zhang, Z. Zhou, X.Y. Lu, M. Xia, *Ceram. Int.*, 2022, 48, 29547-29553.
- T. Tan, S.W. Wang, J.Y. Su, W.H. Yuan, H.Y. Wu, R. Pang, J.T.
 Wang, C.Y. Li, H.J. Zhang, Acs Sustain. Chem. Eng., 2022, 10, 3839-3850.
- 28 F. Zhao, H. Cai, Z. Song, Q. Liu, Chem. Mater., 2021, 33, 3621-3630.
- 29 Z. Jia, C. Yuan, Y. Liu, X.J. Wang, P. Sun, L. Wang, H. Jiang, J. Jiang, *Light Sci. Appl.*, 2020, *9*, 86.
- 30 Q. Wang, Z. Mu, S. Zhang, Q. Zhang, D. Zhu, J. Feng, Q. Du, F. Wu, J. Lumin., 2019, 206, 618-623.
- 31 M. Zhao, S.Q. Liu, H. Cai, F.Y. Zhao, Z. Song, Q.L. Liu, Sci. China Mater., 2022, 65, 748-756.
- 32 S. Fang, T. Lang, T. Han, M. Cai, S. Cao, L. Peng, B. Liu, Y. Zhong, A.N. Yakovlev, V.I. Korepanov, *J. Mater. Chem. C*, 2020, *8*, 6245-6253.
- 33 Q. Wang, Z. Liang, J. Luo, Y. Yang, Z. Mu, X. Zhang, H. Dong, F. Wu, *Ceram. Int.*, 2020, 46, 5008-5014.
- 34 Q. S. Wu, Y. J. Xie, F. She, Q. Zhao, J. Y. Ding, J. C. Zhou, J. Rare Earth., 2021, 39, 1040-1048.
- 35 Y. Fu, P. Xiong, X. Liu, X. Wang, S. Wu, Q. Liu, M. Peng, Y. Chen, J. Mater. Chem. C, 2021, 9, 303-312.
- 36 L. Zhang, D. Wang, Z. Hao, X. Zhang, G.h. Pan, H. Wu, J. Zhang, Adv. Opt. Mater., 2019, 7. 1900185.
- 37 T.C. Lang, M.S. Cai, S.Q. Fang, T. Han, S.S. He, Q.Y. Wang, G.H. Ge, J. Wang, C.Z. Guo, L.L. Peng, S.X. Cao, B.T. Liu, V.I. Korepanov, A.N. Yakovlev, J.B. Qiu, *Adv. Opt. Mater.*, 2022, 10. 2101633.
- 38 D. Liu, X. Yun, P. Dang, H. Lian, M. Shang, G. Li, J. Lin, Chem. Mater., 2020, 32, 3065-3077.
- 39 N. Ma, W. Li, B. Devakumar, Z. Zhang, X. Huang, Mater. Today Chem., 2021, 21. 100512.
- 40 E. Song, X. Jiang, Y. Zhou, Z. Lin, S. Ye, Z. Xia, Q. Zhang, *Adv. Opt. Mater.*, 2019, *7*. 1901105.
- 41 H.Z. Zhang, J. Zhang, Y.C. Su, X.M. Zhang, *Adv. Opt. Mater.*, 2022, *10*. 2200150.